



# Radiochemistry Webinars

## Nuclear Materials Analysis: Physical Methods of Characterization



*In Cooperation with our University Partners*



UNIVERSITY of CALIFORNIA • IRVINE



# Meet the Presenter...

*Dr. Jeff Terry*



Dr. Jeff Terry is a professor of physics at the Illinois Institute of Technology, where his main research focus is on energy systems. His group works to develop new ways of dealing with radioactive waste, understand radiation damage mechanisms in materials, and synthesize novel materials for energy storage and conversion. He also simulates the economic costs of building new energy systems, including small modular nuclear reactors. Prior to joining the faculty at Illinois Tech, he was a staff scientist at Los Alamos National Laboratory (LANL). There, he worked on the Stockpile Stewardship and Management Program and the Waste Isolation Pilot Plant (WIPP) and was a member of the LANL team that sent the first waste shipment to WIPP. He is a former scientific director of the Advanced Test Reactor National Scientific User Facility. Dr. Terry is very active in science communication. He writes a regular column for the Bulletin of the Atomic Scientists and is an editor for the journal Applied Surface Science. Dr. Terry earned his doctorate in chemical physics from Stanford University in 1997 and his bachelor's degree in chemistry from the University of Chicago in 1990.

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# Physical Methods of Characterization of Advanced Nuclear Materials

Dr. Jeff Terry  
Illinois Institute of Technology



National Analytical Management Program (NAMP)  
U.S. Department of Energy Carlsbad Field Office

TRAINING AND EDUCATION SUBCOMMITTEE



# Nuclear Materials Science

- PWR Sustainability
  - Average age of 439 operating nuclear reactors is 24 years
  - Average age of 119 shutdown reactors 22 years
  - The Bulletin of the Atomic Scientists 2008 Status Report
- Push Lifetime To 40, 60, 90 Years
  - Aging
  - Stress
  - Corrosion
  - Fatigue
  - Irradiation effects

# Nuclear Materials Science

- Beyond design basis
  - High temperature
  - Fuel cladding reactions
  - Hydrogen generation

# Nuclear Materials Science

- Gen IV
  - Reactor design
  - Materials issues
  - Corrosion
  - Neutronics
  - High temperature
- Fuel
  - Corrosion
  - Pellet cladding interaction
  - Fission gas release and swelling
  - Hydrogen induced embrittlement of cladding alloys

# Nuclear Materials Science

- Nuclear Waste
  - Fate
  - Transport
  - Geochemistry
  - Biochemistry
  - Repository Science
- Fundamental Science
  - 5f Electron Behavior
  - Highly correlated electrons
  - 5f Chemistry

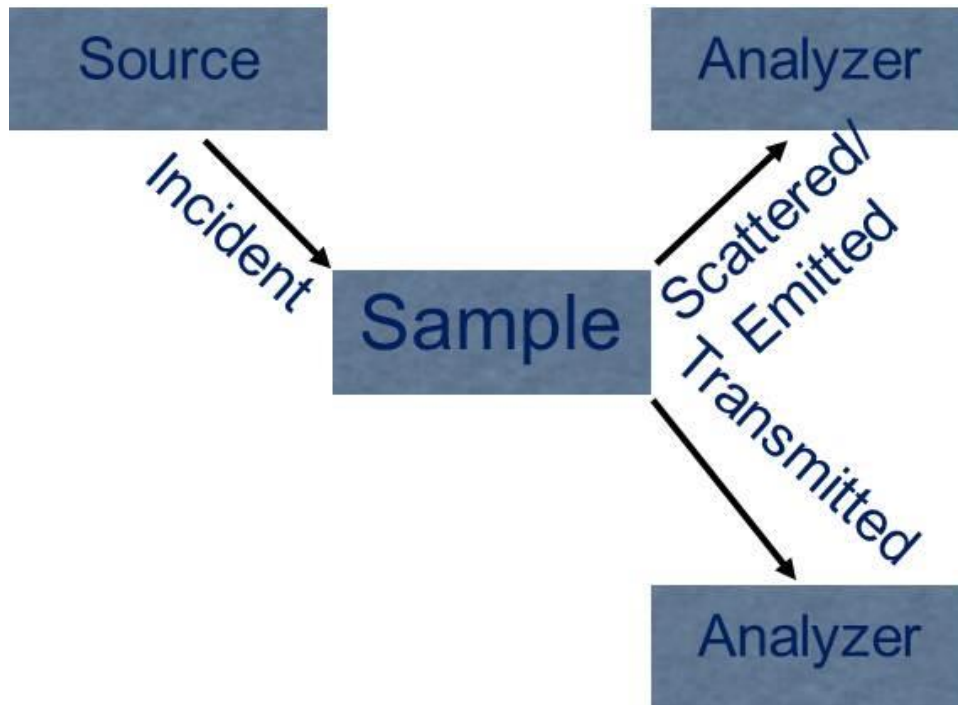
# Techniques

- **Electronic Structure**
  - Photoelectron Spectroscopy
  - X-ray Absorption Near Edge Spectroscopy
  - Tunneling Spectroscopy
  - X-ray Emission Spectroscopy
- **Geometric Structure**
  - Extended X-ray Absorption Fine Structure
  - X-ray Scattering
  - Electron Microscopy
  - Scanning Probe Microscopy
  - Atom Probe
- **Physical Structure**
  - Tomography



# Probes

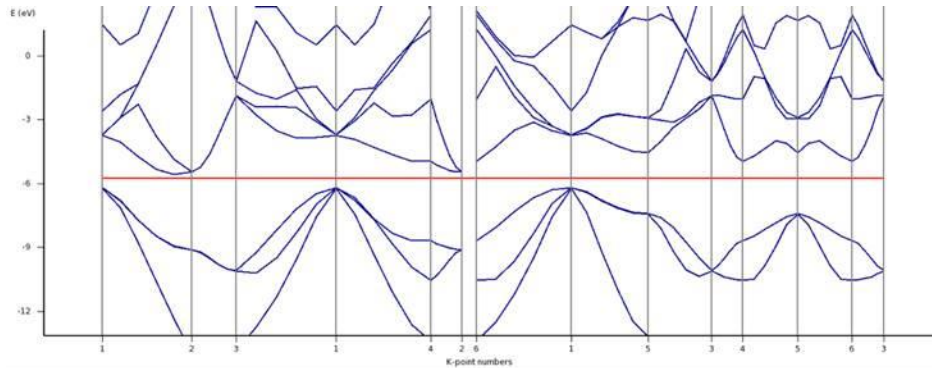
- Scattering Probes



In	Out	Name
photon	0	Absorption
photon	photon	Fluorescence, Raman, Compton
photon	electron	Photoelectron Spectroscopy
photon	electron *electron	Auger
electron	electron *electron	Auger
electron	photon	X-ray Emission
electron	electron	Energy Loss
electron	electron	Microscopy
ion	ion	Secondary Ion Mass Spectroscopy

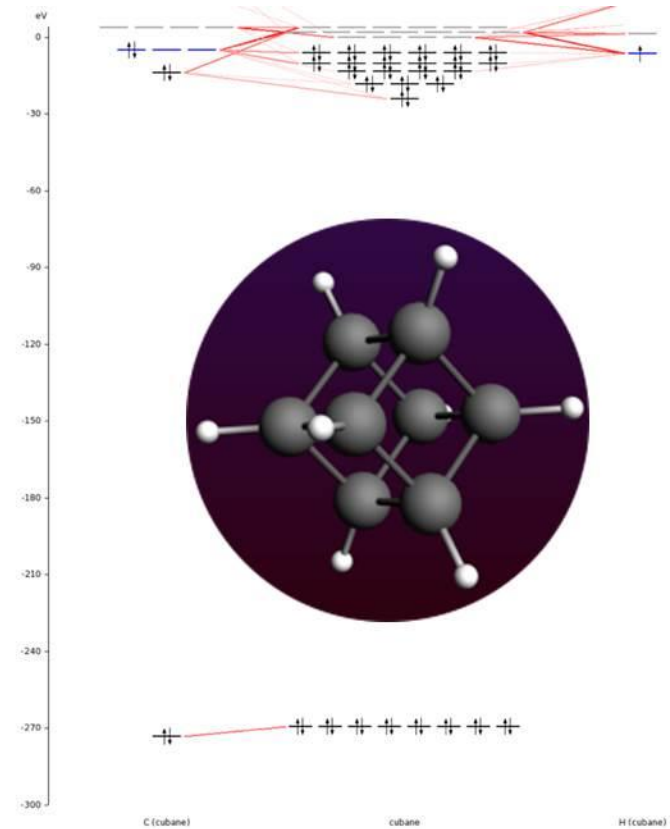
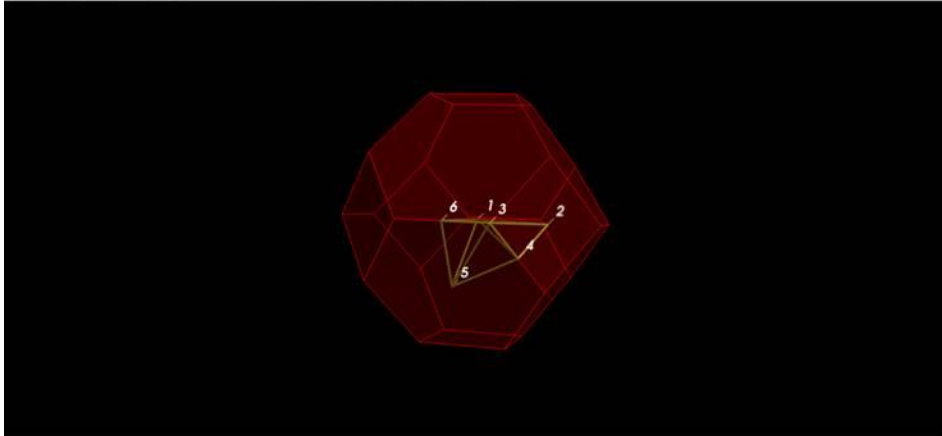
# Electronic Structure

- Chemical Environment - Energy Position
- Physical Environment - Broadening



Brillouin Zone Path Visualisation

Show K Points  
  Show Lattice Vectors  
  Show Fermi Surface  
  Show spin surface  
  Show Brillouin Zone  
  Show Path

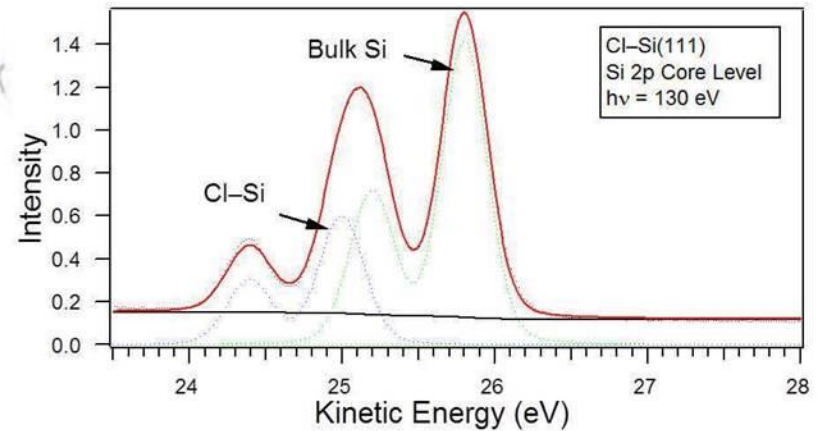
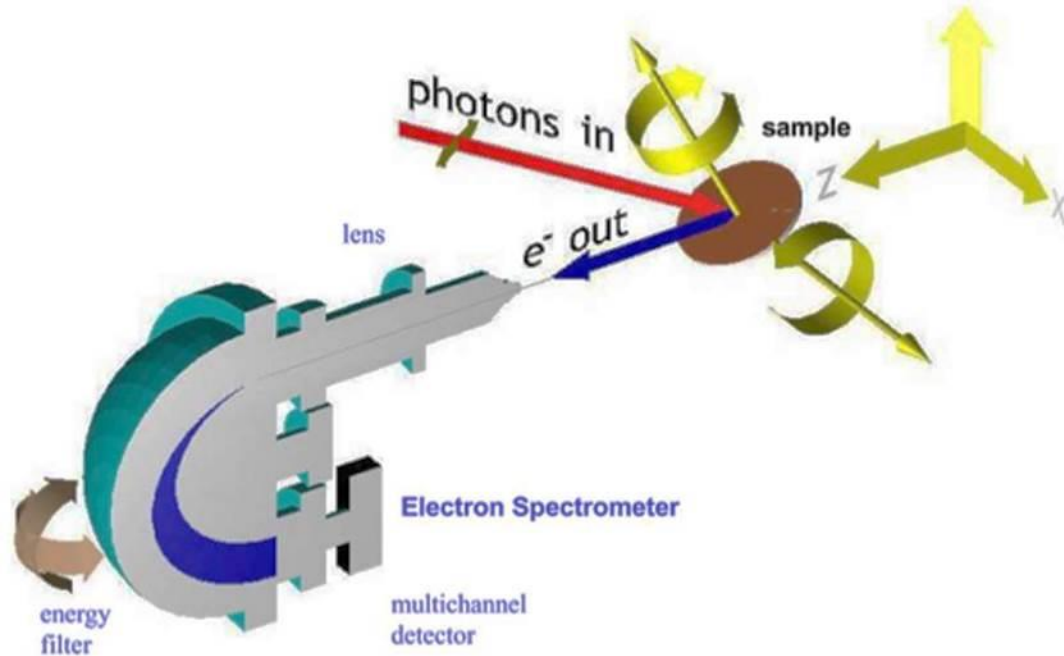


C (cubane)

cubane

H (cubane)

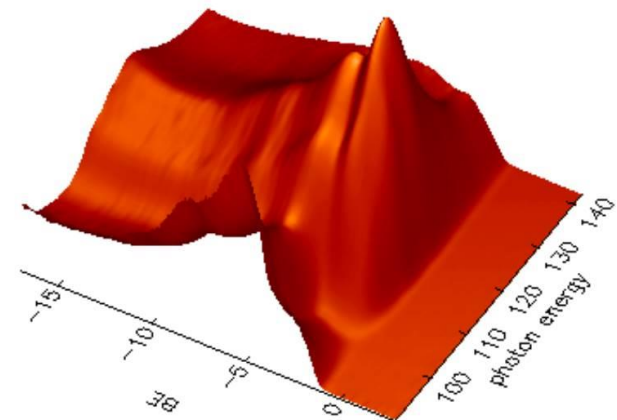
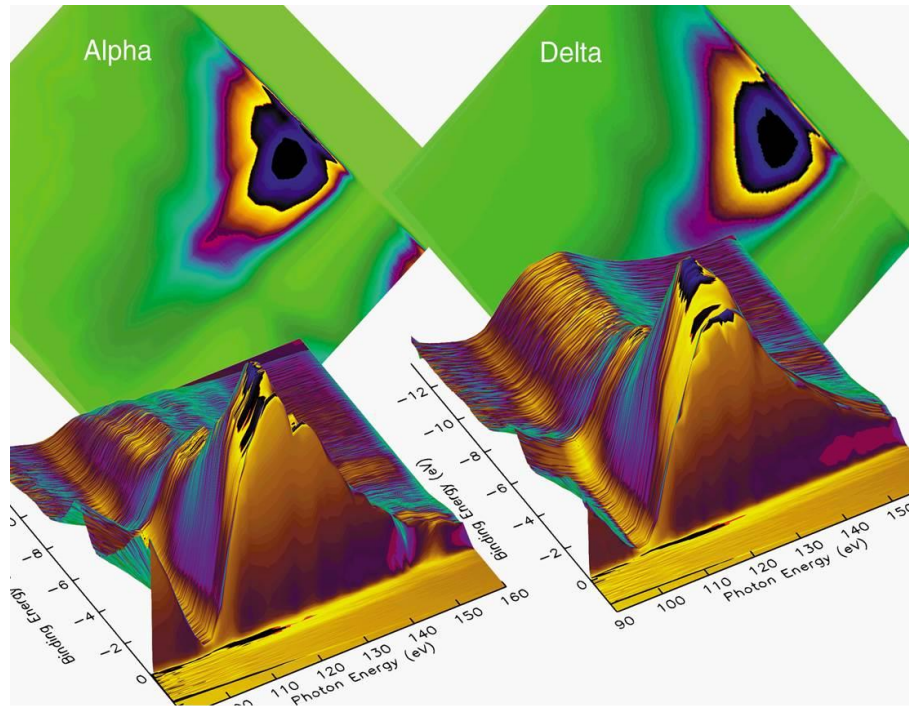
# Photoelectron Spectroscopy



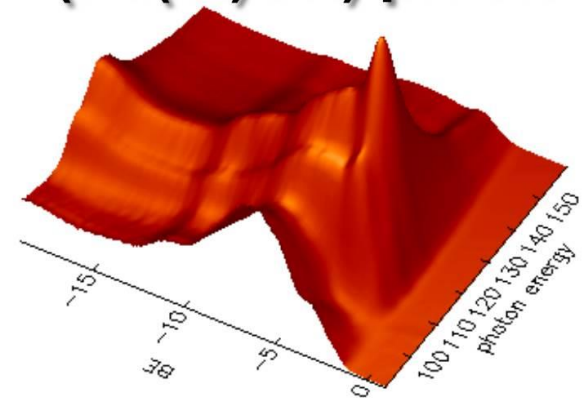
From Rotenberg: [www-bl7.lbl.gov/BL7/who/eli/SRSchoolER.pd](http://www-bl7.lbl.gov/BL7/who/eli/SRSchoolER.pd)

# Resonant Photoemission of Pu

- 5f Electron systems are different



**(Pu(VI)OH) pH 5.5**



**Pu(V) on SiO<sub>2</sub>/Si**

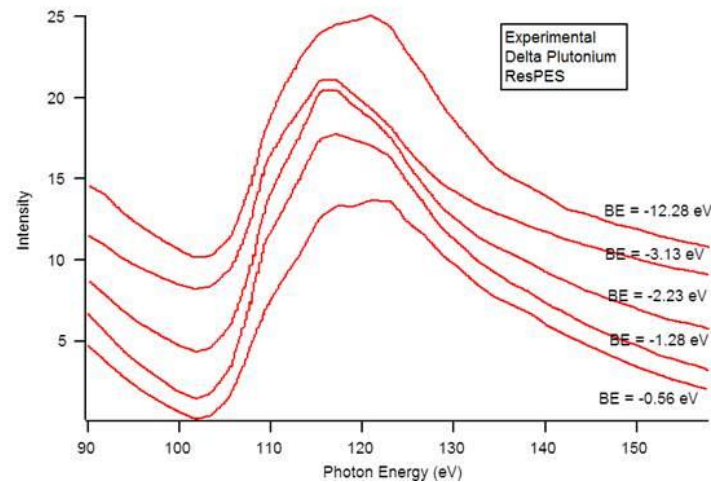
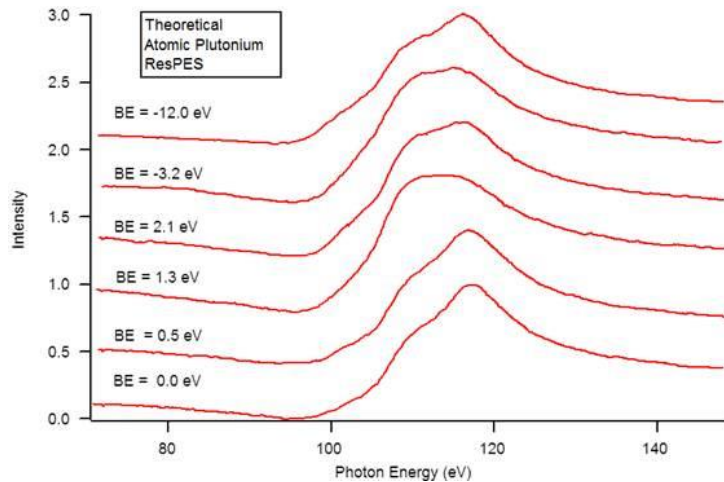
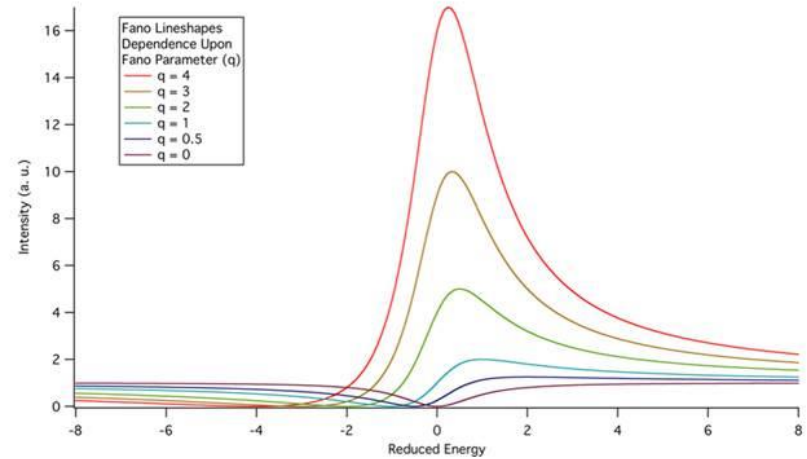
Terry, J, et al., Surface Science 499, L141 (2002)

Tobin, J. G, et al., Physical Review B 68, 155109 (2003)

# Resonant Photoemission of Pu

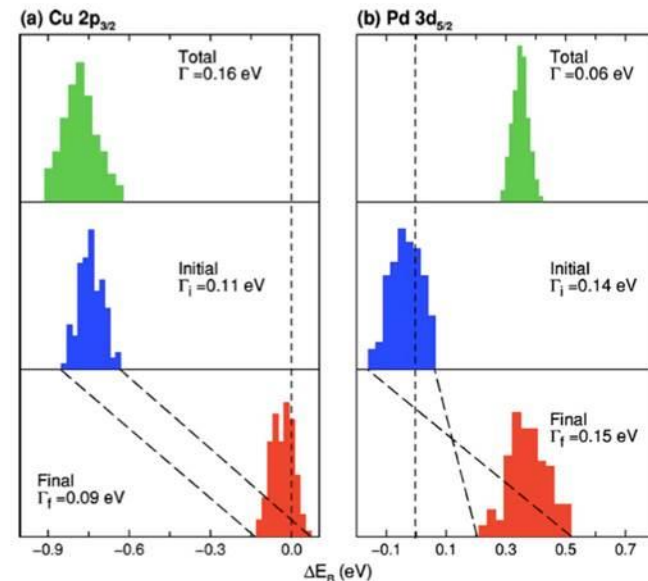
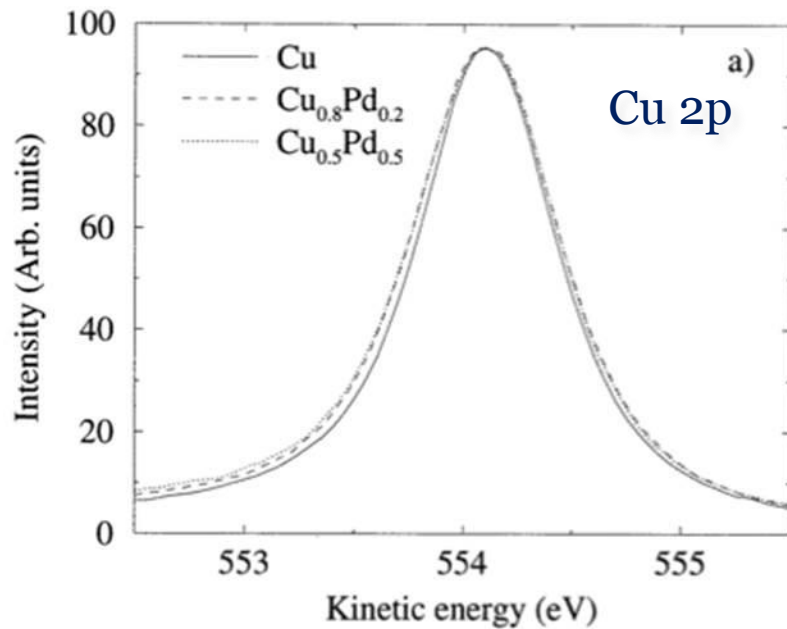
- Fano profile
- FCC Pu suggests more atomic-like character

$$q \sim \langle 5d^{10}5f^5 | r | 5d^95f^6 \rangle / \langle 5d^{10}5f^5 | r | 5d^{10}5f^4 + e^- \rangle$$



# Photoemission Of Disordered Alloys

- High resolution core level spectroscopy allows one to measure disorder broadening
- Can now calculate by calculating the core level shift for every atom



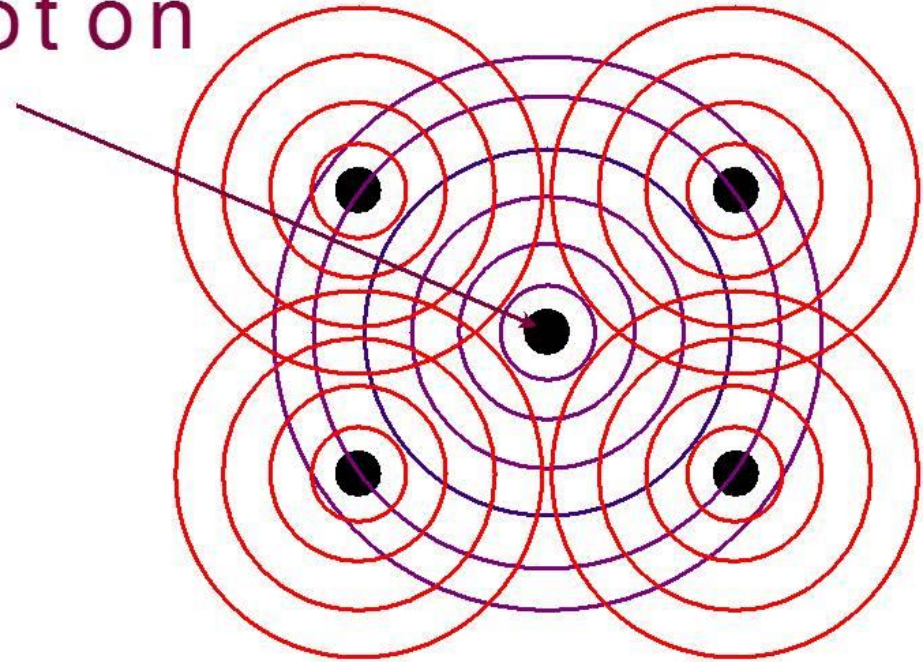
Cole, et al., J. Phys.: Condens. Matter 10 (1998) 5679–5695.

Martin, et al, PHYSICAL REVIEW B 72, 054210 (2005)  
 W. Olovsson et al. / Journal of Electron Spectroscopy and Related  
 Phenomena 178–179 (2010) 88–99

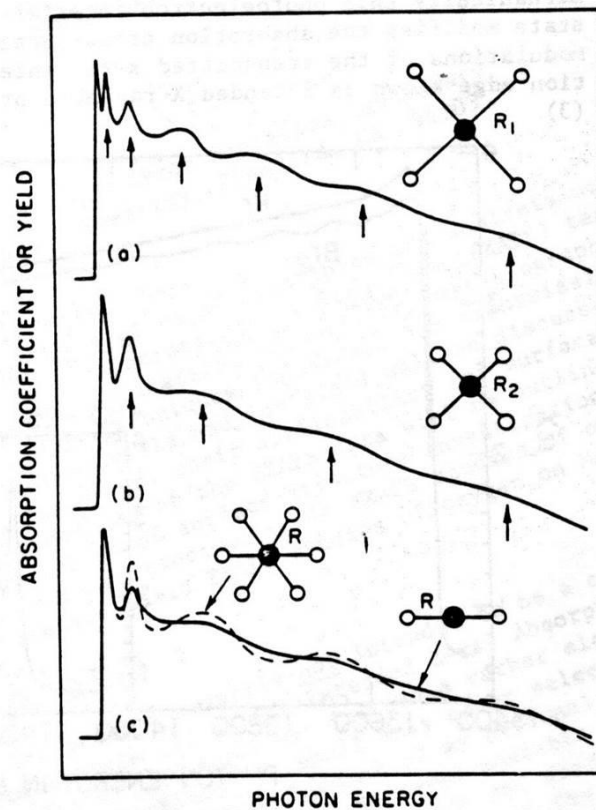
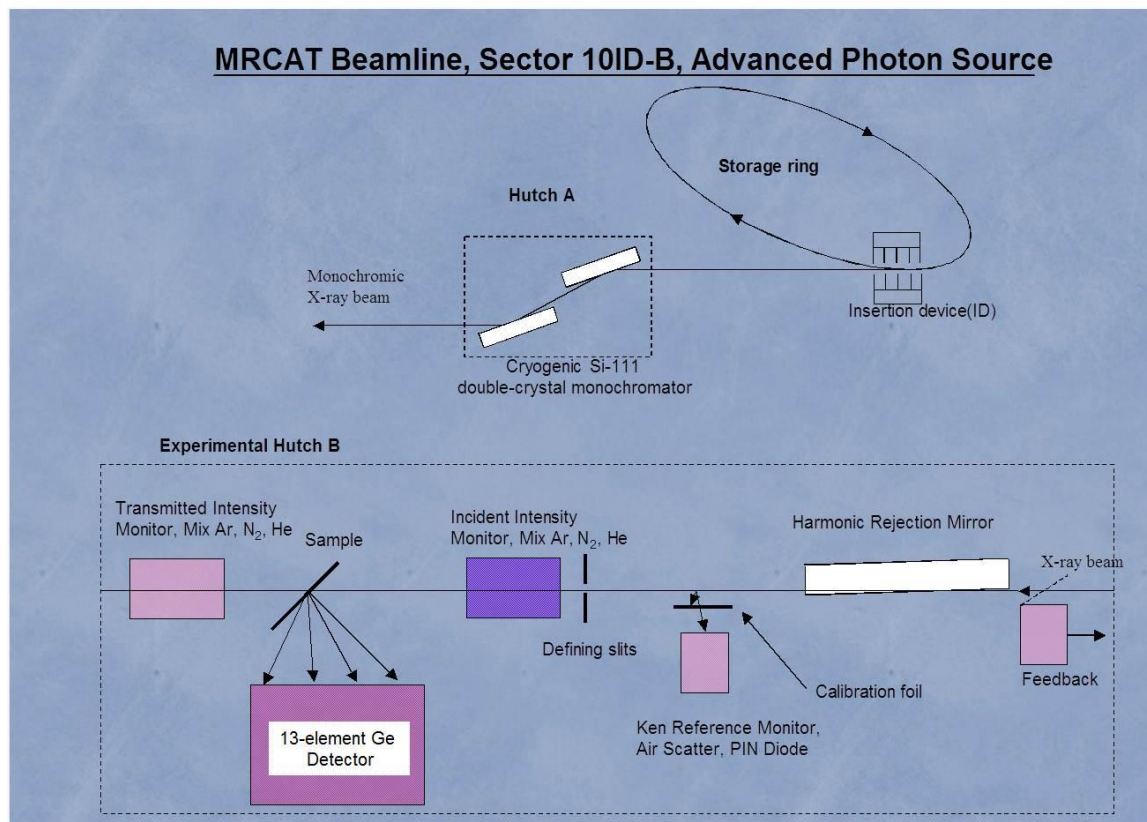
# X-ray Absorption

- Photon In
- Electron Excite
- Electron Scatters
- Electron Interference

Photon

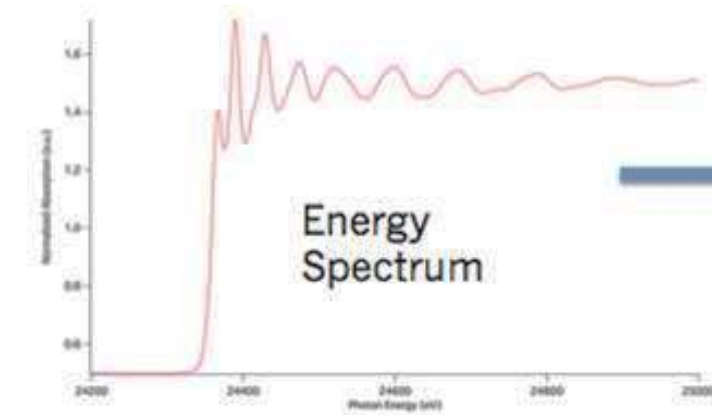


# X-ray Absorption



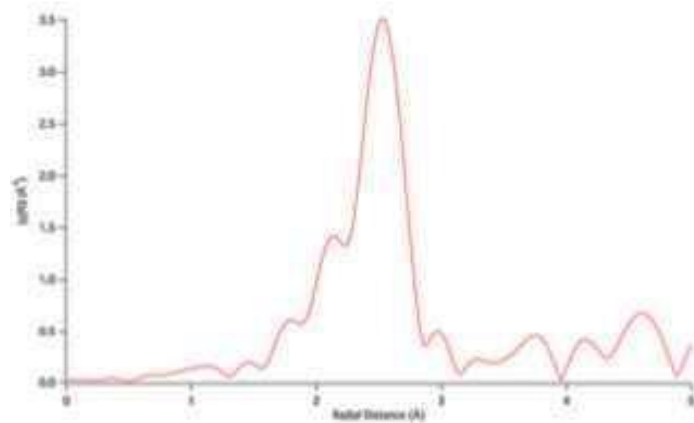
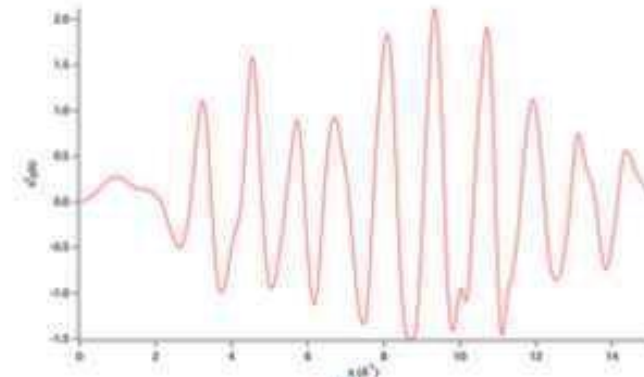


# XAS Analysis



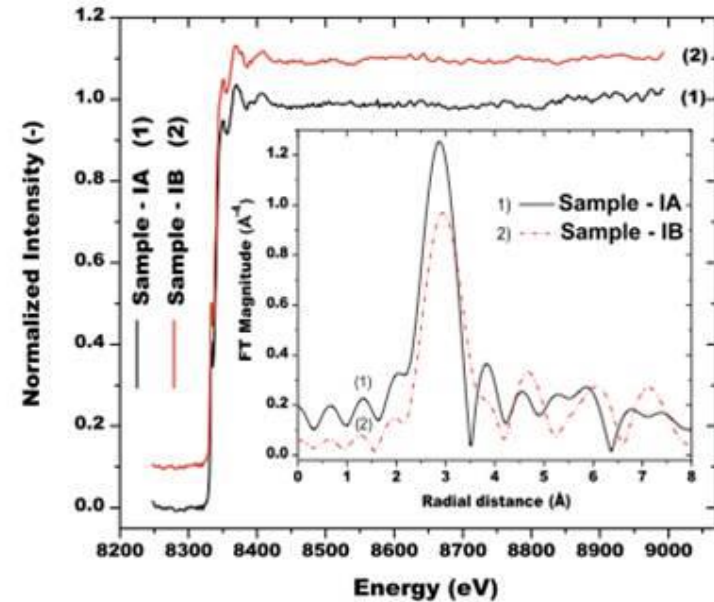
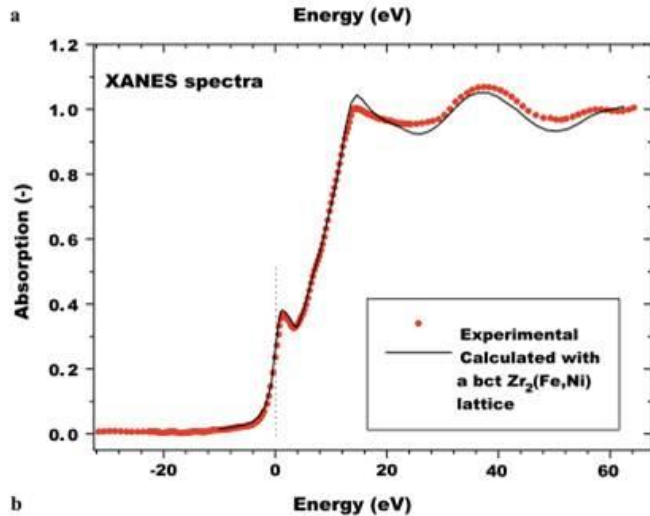
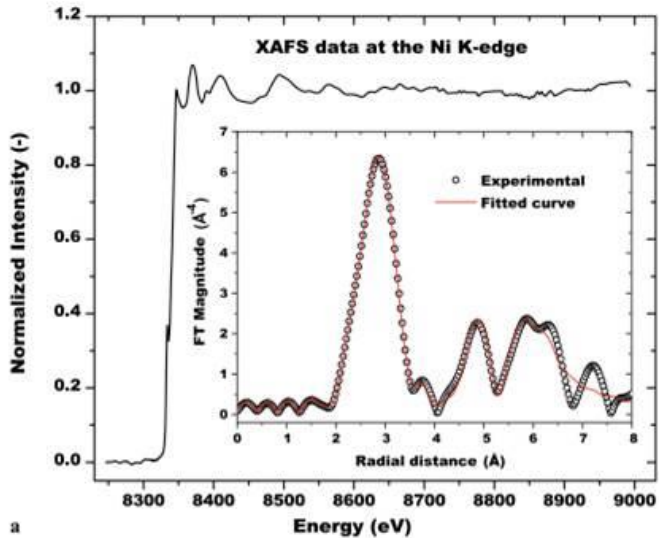
Convert to k-space and extract oscillations as a function of wavenumber

$$k = \sqrt{\frac{2m(E - E_0)}{\hbar^2}}$$



Fourier transform into R-space

# Irradiated Zircaloy-2

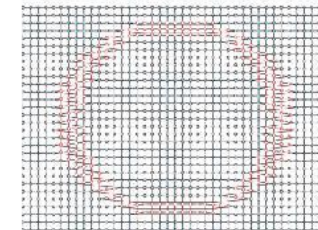
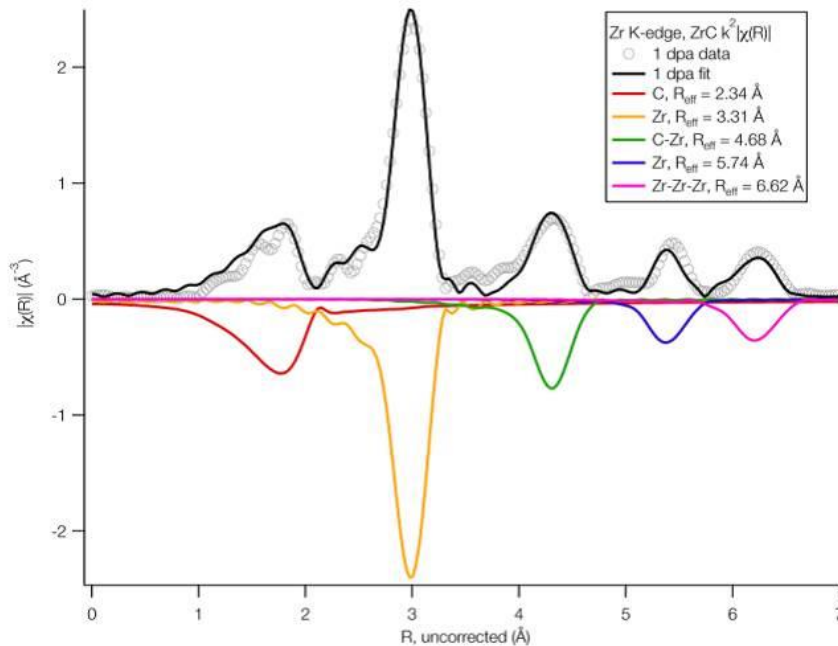


1A -  $5.9 \times 10^{21} \text{ n/cm}^2$

1B -  $14.2 \times 10^{21} \text{ n/cm}^2$

- $\text{Zr}_2\text{Fe}_{0.6}\text{Ni}_{0.4}$
- Dissolution and disorder of secondary phase

# ZrC as Cladding



Olive, et al., Journal of Nuclear Materials 475, 123 (2016)

Fitting results for the simultaneous fit of the 0 and 1 dpa ZrC data. A single value of  $E_0$  was used for all paths, the best fit was  $-1.12 \pm 0.62$  eV. Uncertainties determined by inversion of the covariance matrix.

Path <sup>a</sup>	$R_{\text{PerfectCrystal}} (\text{Å})$	$R_{\text{Measured}} (\text{Å})$	$\sigma^2 (\text{Å}^2)$	N, 0 dpa	N, 1 dpa	% Amplitude Difference
Zr <sub>0</sub> → C <sub>1</sub> → Zr <sub>0</sub>	2.34	2.32 ± 0.02	0.008 ± 0.002	7.7 ± 2.6	6.6 ± 1.7	-14%
Zr <sub>0</sub> → Zr <sub>1</sub> → Zr <sub>0</sub>	3.31	3.32 ± 0.02	0.004 ± 0.001	12.9 ± 3.5	9.3 ± 0.5	-28%
Zr <sub>0</sub> → C <sub>1</sub> → Zr <sub>2</sub> → Zr <sub>0</sub>	4.68	4.69 ± 0.02	0.005 ± 0.001	12.0 ± 2.0	7.8 ± 1.2	-35%
Zr <sub>0</sub> → Zr <sub>3</sub> → Zr <sub>0</sub>	5.74	5.75 ± 0.02	0.003 ± 0.001	13.0 ± 3.5	6.0 ± 1.7	-53%
Zr <sub>0</sub> → Zr <sub>1</sub> → Zr <sub>4</sub> → Zr <sub>1</sub> → Zr <sub>0</sub>	6.62	6.60 ± 0.02	0.005 ± 0.001	6.6 ± 1.8	2.8 ± 0.8	-57%

<sup>a</sup> The subscript represents nth coordination shell of a particular type of atom, 0 indicating the core scattering atom.

# ZrC

- Sphere 1.8 nm in diameter
- Model undefected regions as spherical

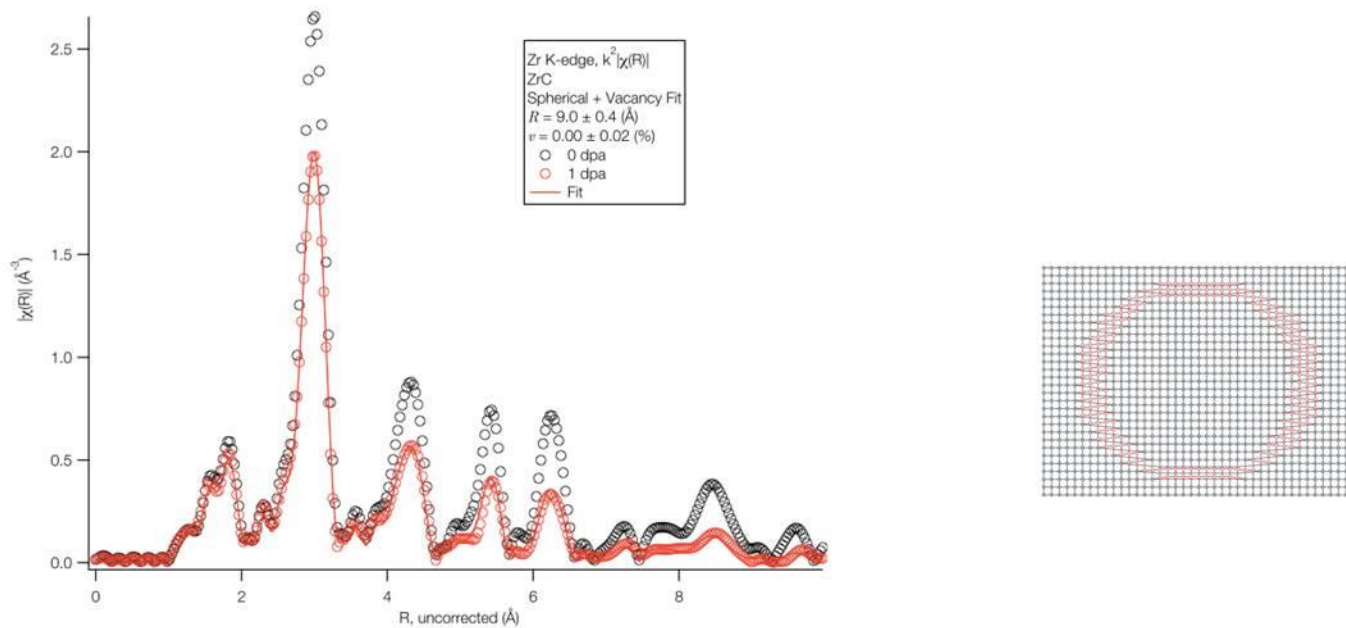
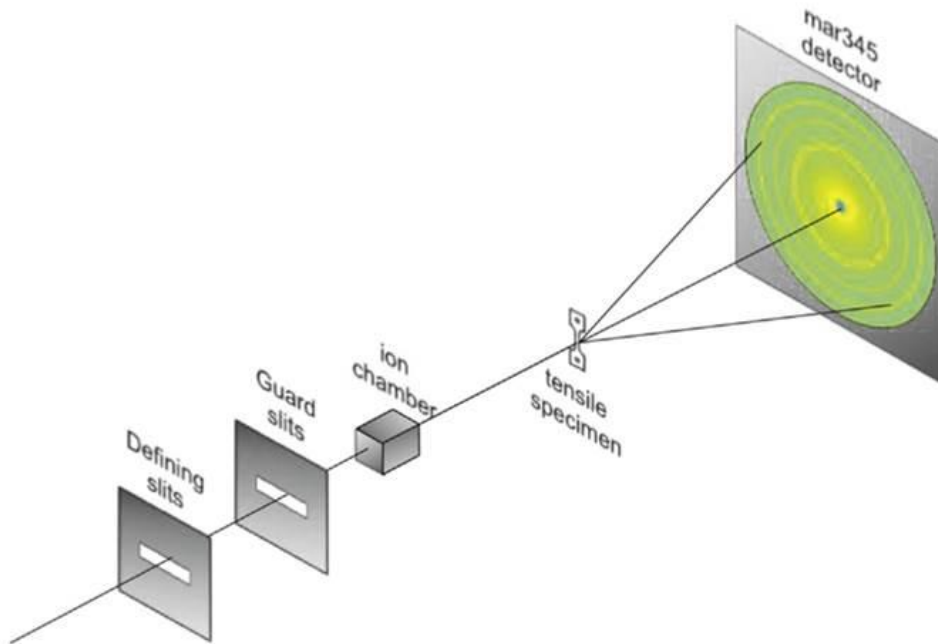


Fig. 8. Fitting results of the unirradiated ZrC sample to the 1 dpa sample, using the vacancy corrected spherical crystallite model.

Olive, et al., Journal of Nuclear Materials 475, 123 (2016)

$$N_{\text{nano}} = \left[ 1 - \frac{3}{4} \left( \frac{r}{R} \right) + \frac{1}{16} \left( \frac{r}{R} \right)^3 \right] (N_{\text{bulk}}(1 - v))$$

# X-ray Scattering

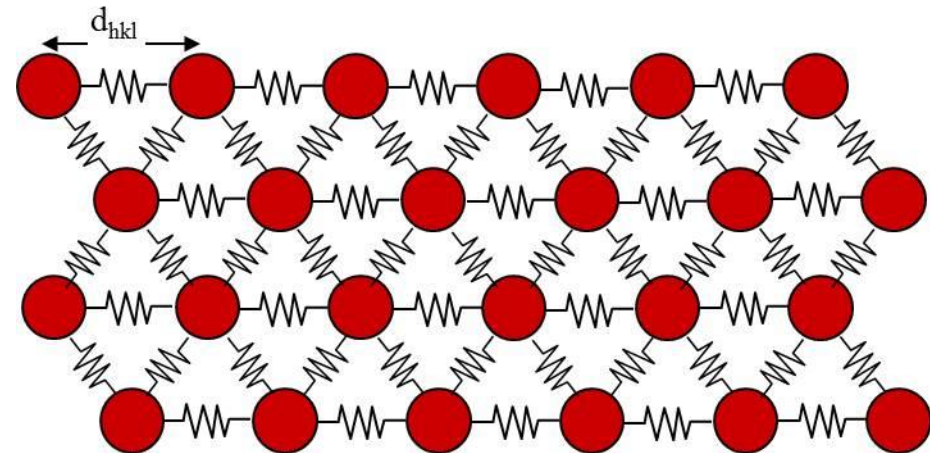


Liu, et al., Materials Science & Engineering A 651, 55 (2016)

- Geometric structure
- Phase identification
- Size/Shape/Stress

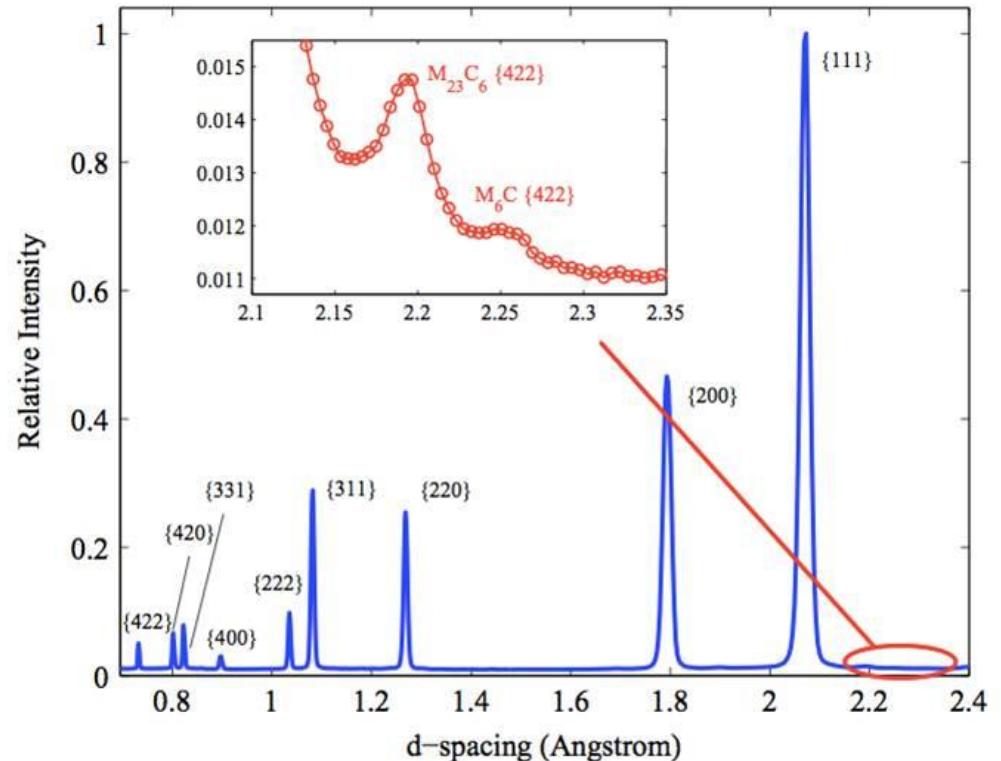
Calculate lattice strains (elastic) from change in atomic spacing.

$$\varepsilon = \frac{d_{hkl} - d_0}{d_0}$$



# Alloy 617: Intermediate Heat Exchanger (IHX) of VHTR

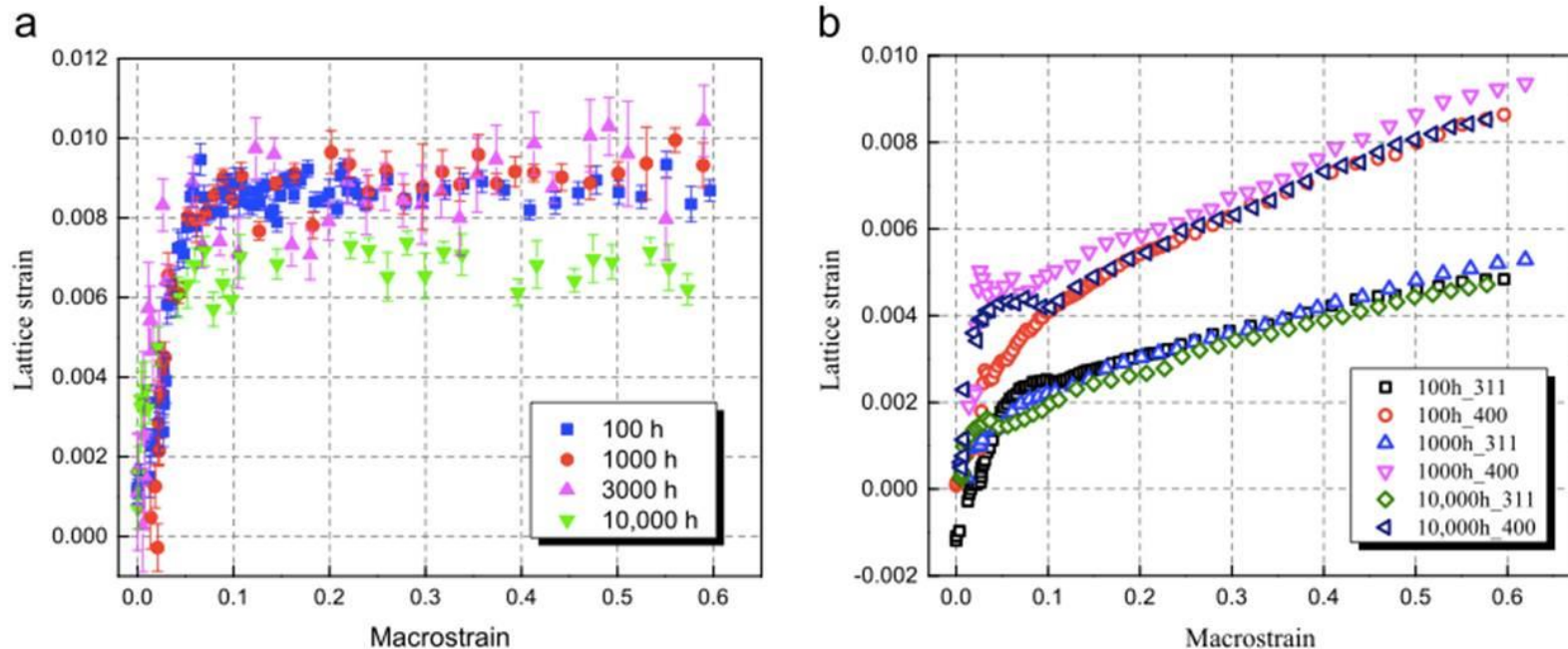
- $M_{23}C_6$  and  $M_6C$
- Carbide phase analysis complicated by containment



**Fig. 3.** The  $d$ -spacing plot of Alloy 617 at  $t=10,000$  h showing {422} reflections from the  $M_{23}C_6$  and  $M_6C$  phases. Major reflections from the face-centered cubic (FCC) matrix have also been indexed.

# Alloy 617: Intermediate Heat Exchanger (IHX) of VHTR

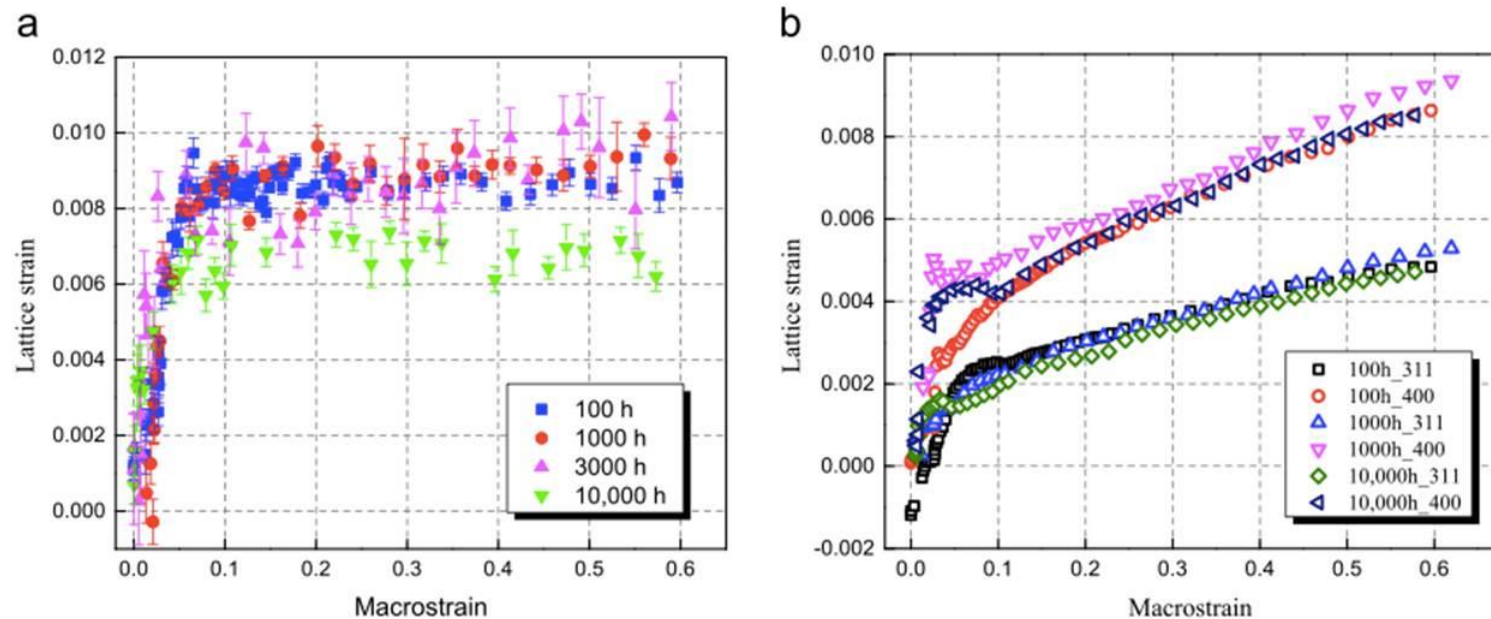
- Initial Linear Increase in  $M_{23}C_6$  Lattice Strain



**Fig. 6.** Lattice strain evolution during the tensile loading: (a) the lattice strains of the  $M_{23}C_6$  precipitate {422} reflection vs the macrostrain, and (b) the lattice strains of the metallic matrix {311} and {400} reflections vs the macrostrain, for Alloy 617 aged at 1000 °C to different times.

# Alloy 617: Intermediate Heat Exchanger (IHx) of VHTR

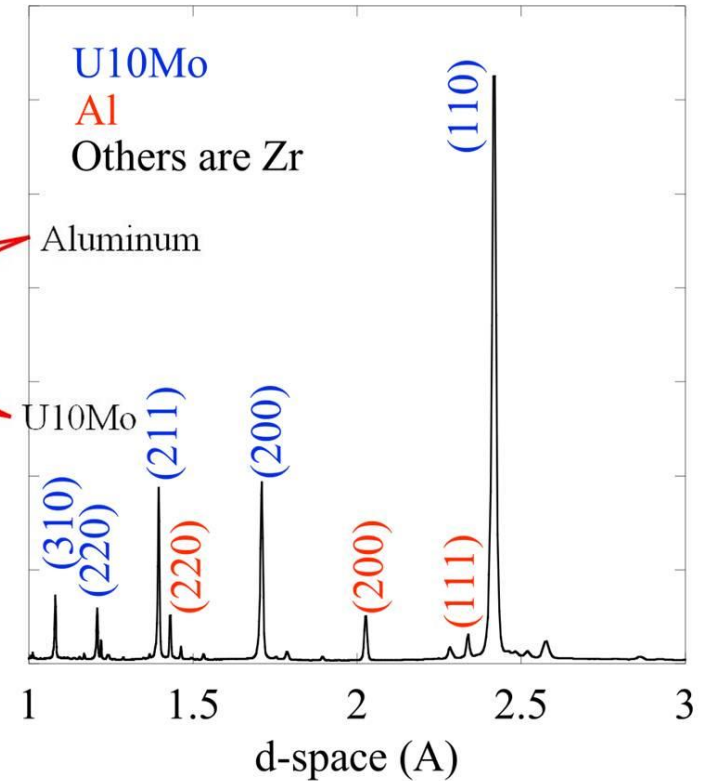
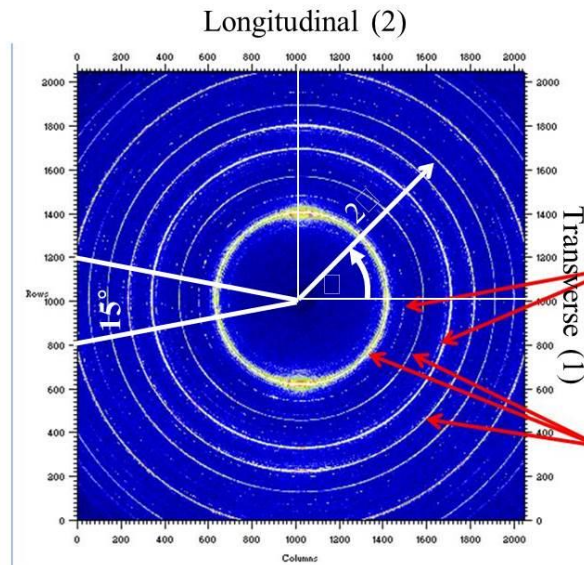
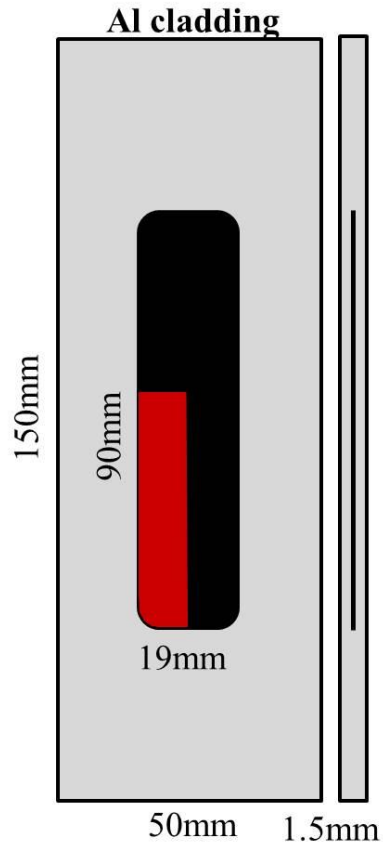
- Becomes constant as nucleation of microcracks or microvoids occur
- Leading to the fracture or decohesion of the particles



**Fig. 6.** Lattice strain evolution during the tensile loading: (a) the lattice strains of the  $M_{23}C_6$  precipitate {422} reflection vs the macrostrain, and (b) the lattice strains of the metallic matrix {311} and {400} reflections vs the macrostrain, for Alloy 617 aged at 1000 °C to different times.



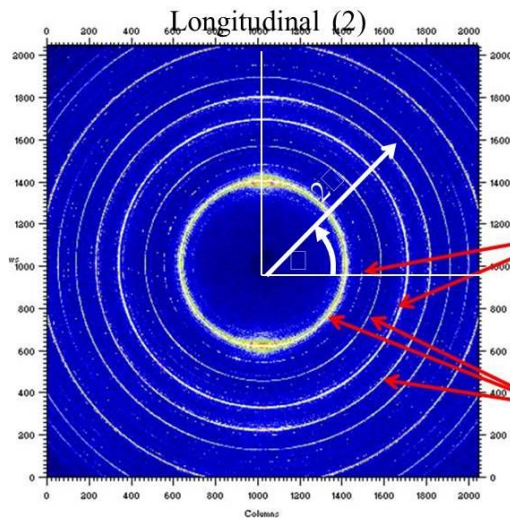
# U10Mo Fuel



1.5 mm total thickness. 0.25mm thick U10Mo foil. Foil was cold rolled with Zr diffusion layer, then HIP bonded.

# U10Mo

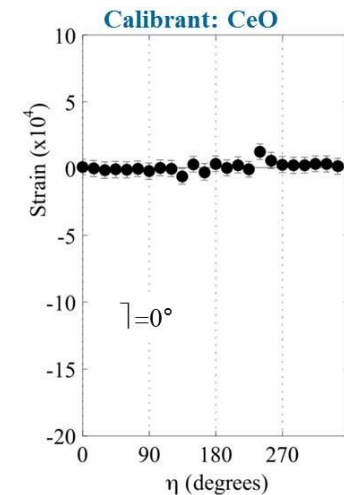
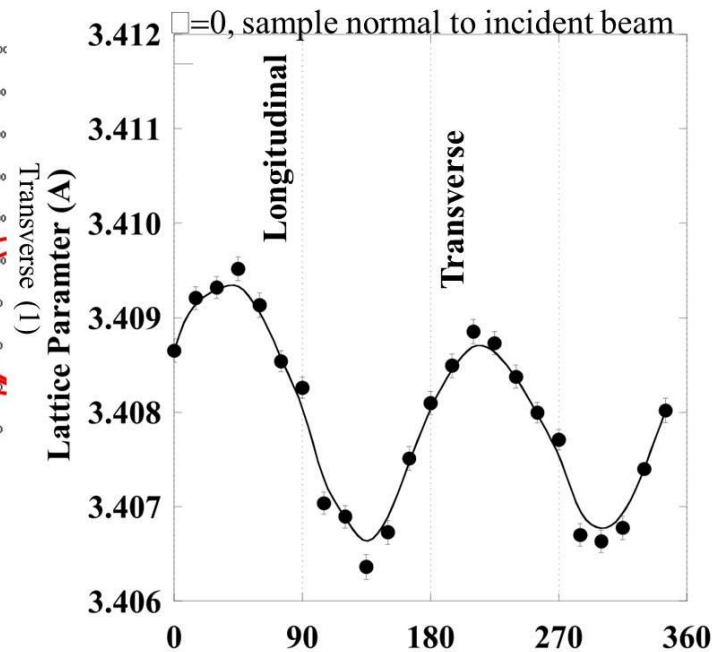
## • Map Full Strain Tensor



$$\varepsilon(\varphi, \psi) = \frac{d(\varphi, \psi) - d_0}{d_0}$$

6 fitting parameters, 72 measurements of  $d(\varphi, \psi)$

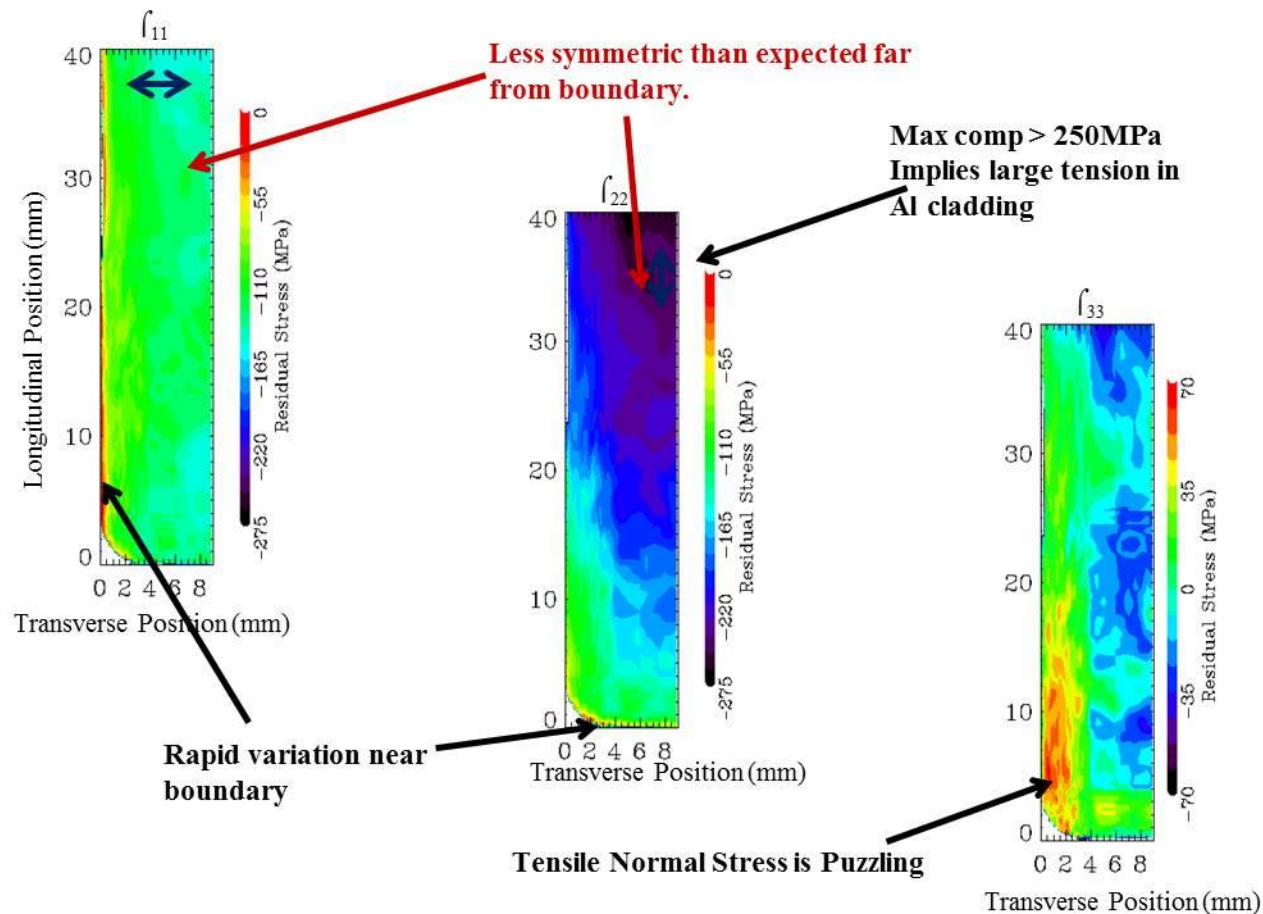
$$= \varepsilon_{11} \cos^2 \phi \sin^2 \psi + \varepsilon_{22} \sin^2 \phi \sin^2 \psi + \varepsilon_{33} \cos^2 \psi + \varepsilon_{12} \sin 2\phi \sin^2 \psi + \varepsilon_{23} \sin \phi \sin 2\psi + \varepsilon_{13} \cos \phi \sin 2\psi$$



$\eta$  Collect data at 3 sample angles,  $0^\circ$ ,  $45^\circ$ , and  $60^\circ$ . Note, we cannot get a diffraction vector normal to the surface, i.e.  $\psi = 90^\circ$

# U10Mo

- Rapid approach to zero stress at boundary suggest the material is not bonded on lateral interface

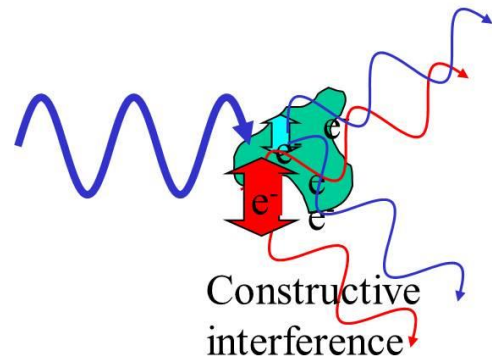


~5000 data points, 72 diffraction patterns per point = ~350K diffraction patterns

# Small Angle X-ray Scattering

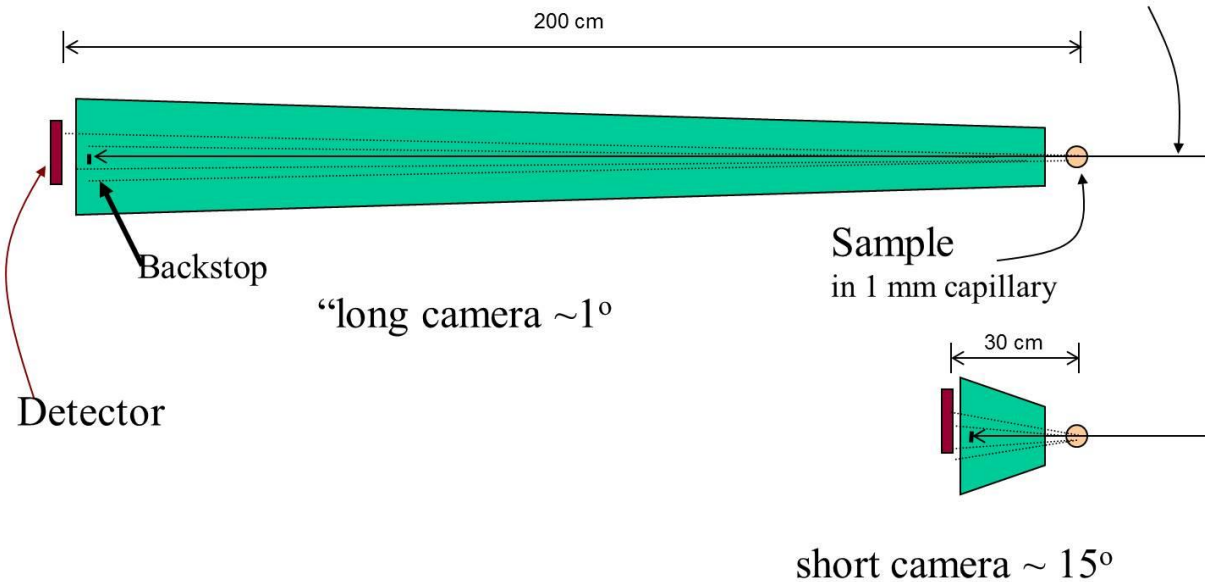
- Sensitive To Shape and Distribution of Electrons

Destructive interference



## Experimental Geometry

Collimated X-ray beam



# SAXS of ODS Steel

- $Y_2Ti_2O_7$  and  $Y_2TiO_5$  nanoparticles at 960 C Diameter 2.6 nm

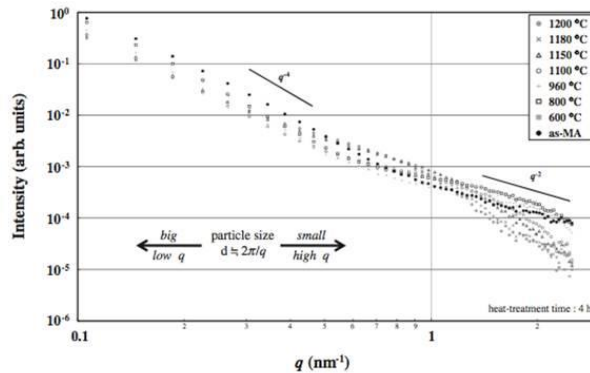


Fig. 3 SAXS profiles of as-MA powder and heat-treatment powders (600, 800, 960, 1100, 1150, 1180 and 1200°C for 4 h).

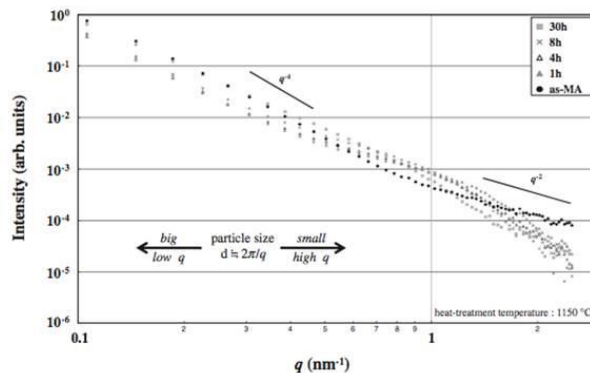


Fig. 4 SAXS profiles of as-MA powder and heat-treatment powders (1150°C for 1, 2, 4, 8, 15 and 30 h).

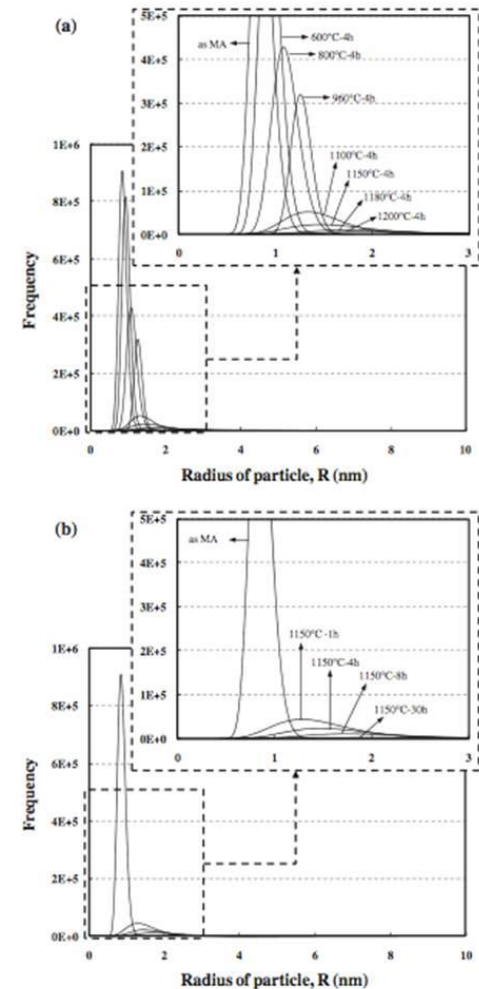
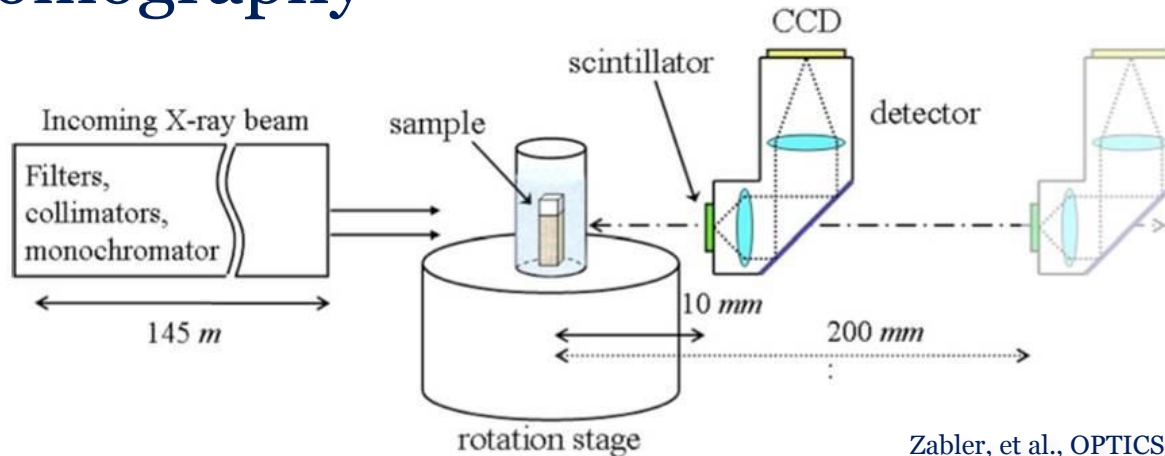


Fig. 5 Results of profile analysis for the distribution of particle size and frequency (a) effects of heat-treatment temperature; (b) effects of heat-treatment time.

# Tomography

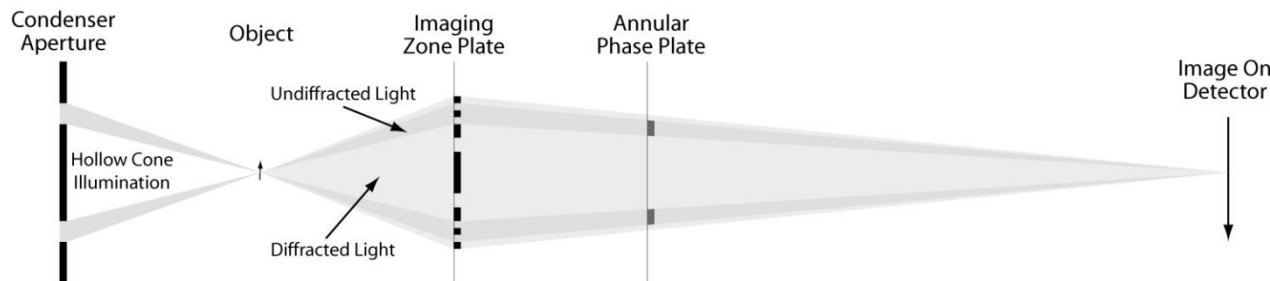
- Microtomography



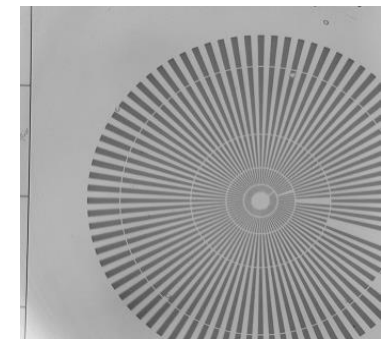
Zabler, et al., OPTICS LETTERS 32, 2987 (2007)

- Nanotomography

Schematic diagram of the Zernike phase contrast X-ray microscope



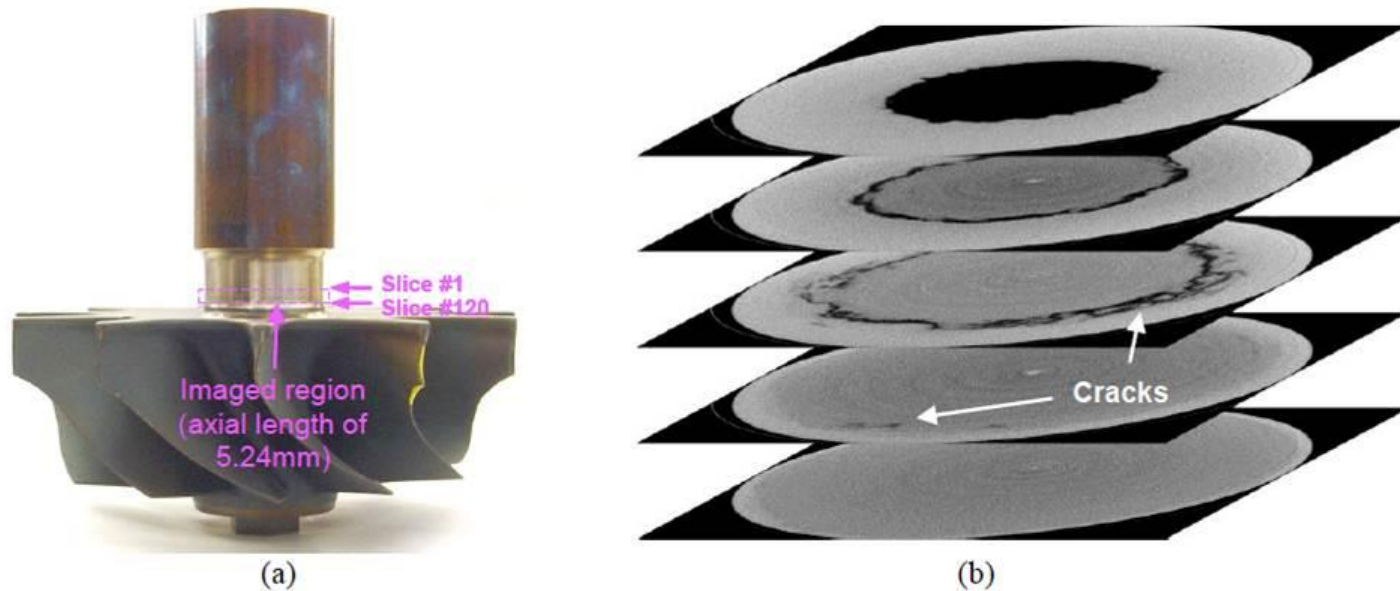
Pianetta unpublished



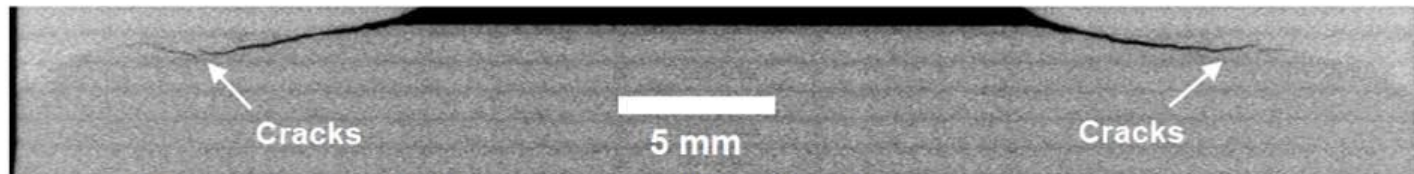
X-ray transmission image of the Siemens calibration standard with 30 nm minimum features. FOV: 30x30 microns<sup>2</sup>

# Friction Weld Crack TiAl/Ti<sub>6</sub>Al<sub>4</sub>V

- 150 keV at 50  $\mu\text{m}$  Resolution



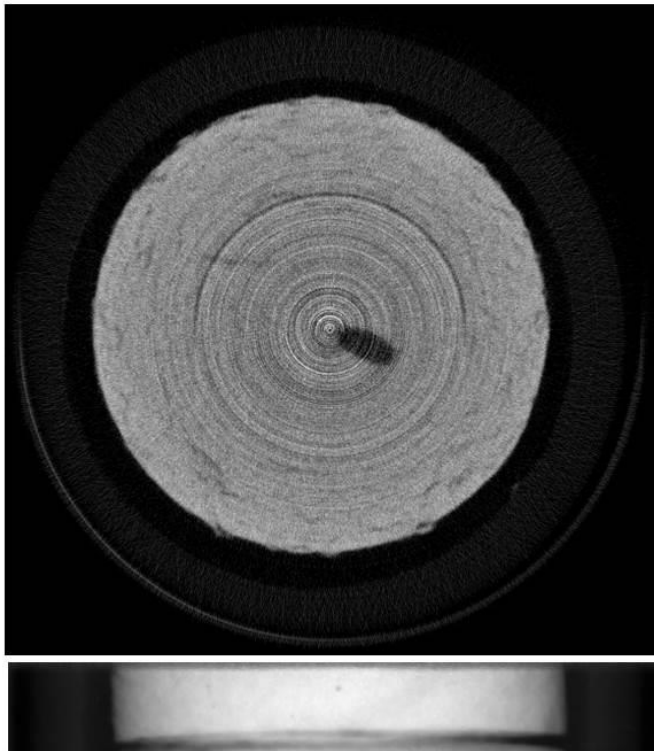
**FIGURE 6.** (a) Welding sample #2 and (b) typical reconstructed X-ray CT slices around the welding interface (distance between two adjacent slices is 0.3 mm).



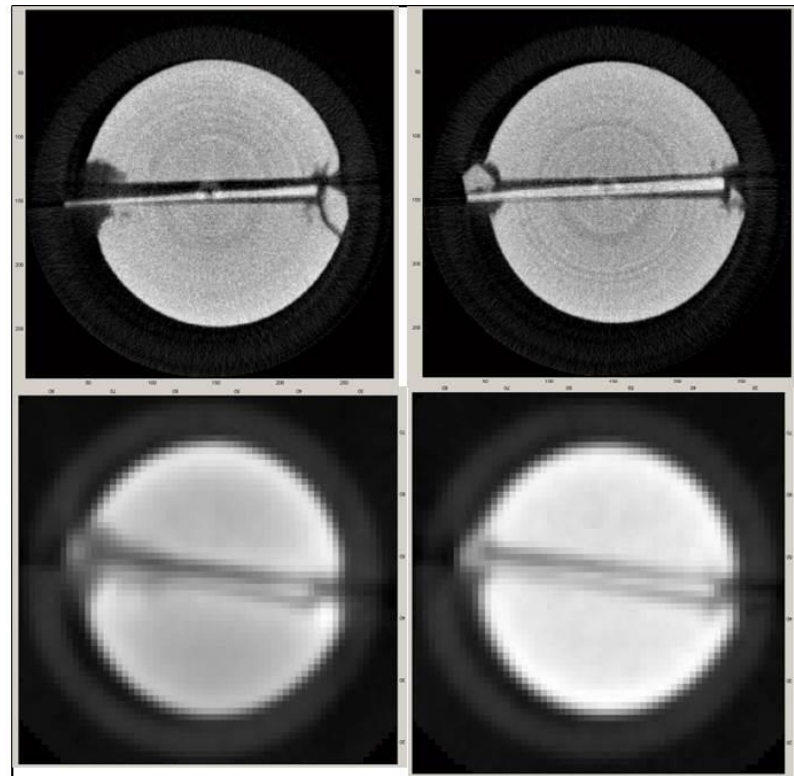
**FIGURE 7.** A re-sliced axial CT image of the weld sample #2 shown in Fig. 6a.

# Tomography of Fuel Pin

- High Energy Tomography Of  $\text{UO}_2$  Fuel In a Steel Clad Fuel Pin
  - White Beam Filtered To 251 KeV
  - Cadmium-Tungstate Scintillator - 5  $\mu\text{m}$  Resolution



Bourke, et al., in preparation

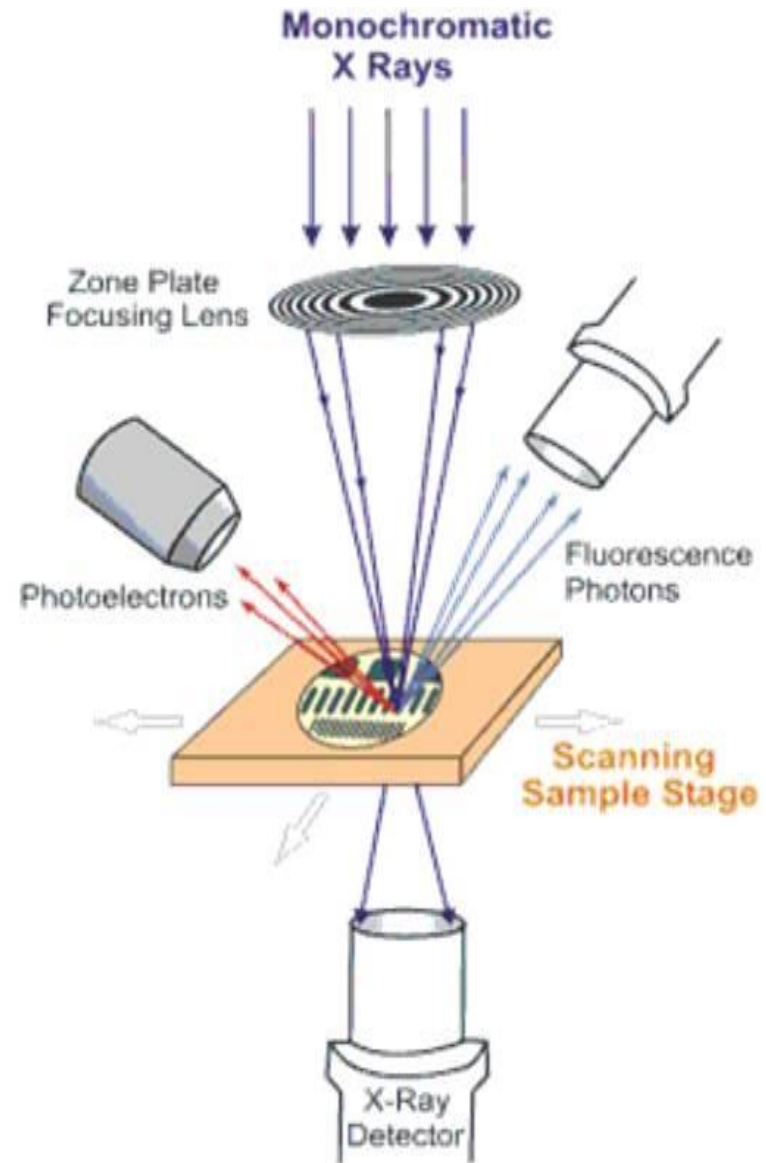


251 KeV Synchrotron Data (Top) vs LANL 450 KeV Brehmstrahlung Source



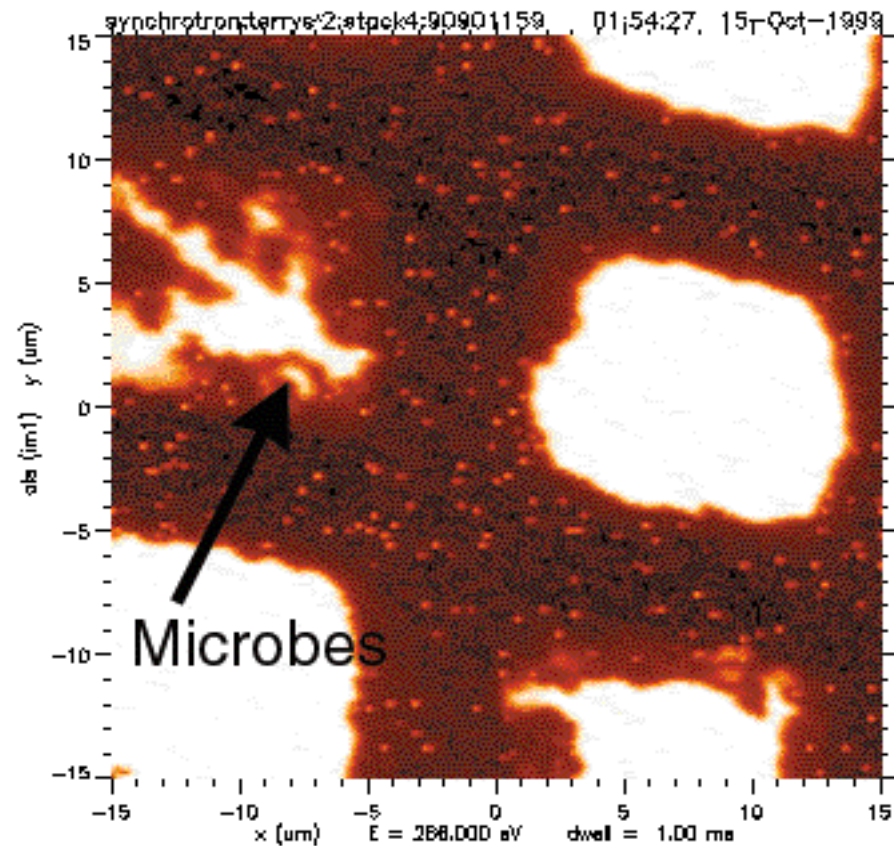
# Mapping

- All the aforementioned techniques can be combined with mapping by using the spacial profile of the X-ray beam



# Pu(VI) Sorption on WIPP 1A Bacteria

- C K-edge Map
- 100 nm Resolution

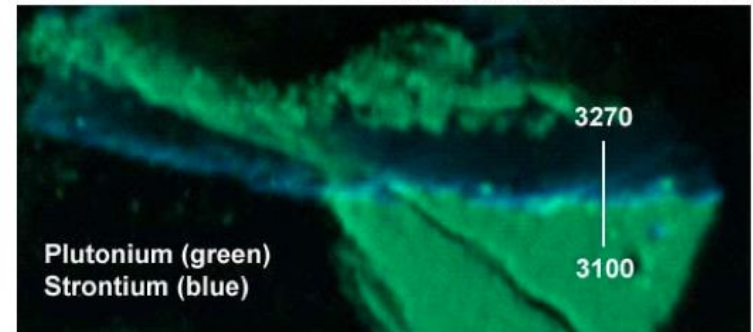
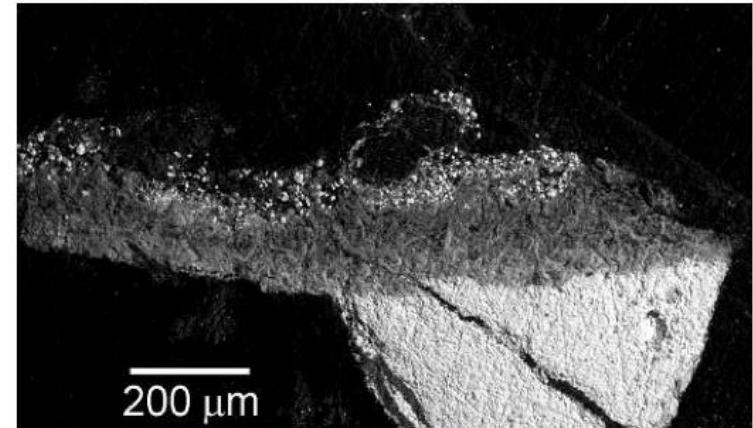


# Corrosion of Commercial Spent Fuel

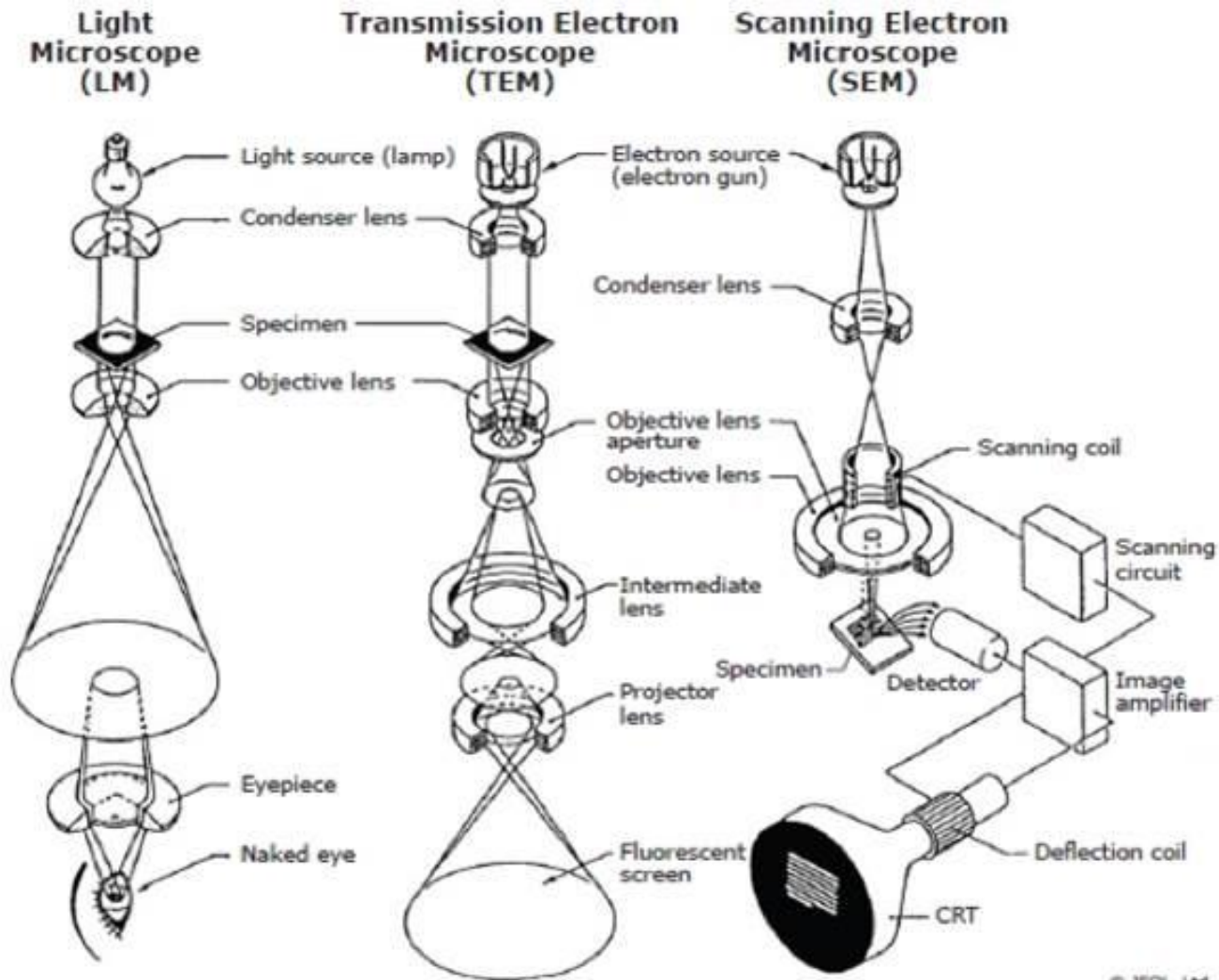
- Fluorescence map
- Only spent fuel sample at synchrotron???
- [Np] drops with 20 microns from 1:1500 to 1:7000 proportional U 4+/6+ fraction



Figure 2.2. Miniature Drill Press for Coring CSNF Specimens Using a Hollow Stainless Steel Bit with Diamond Abrasive, Shown in the Hot Cell



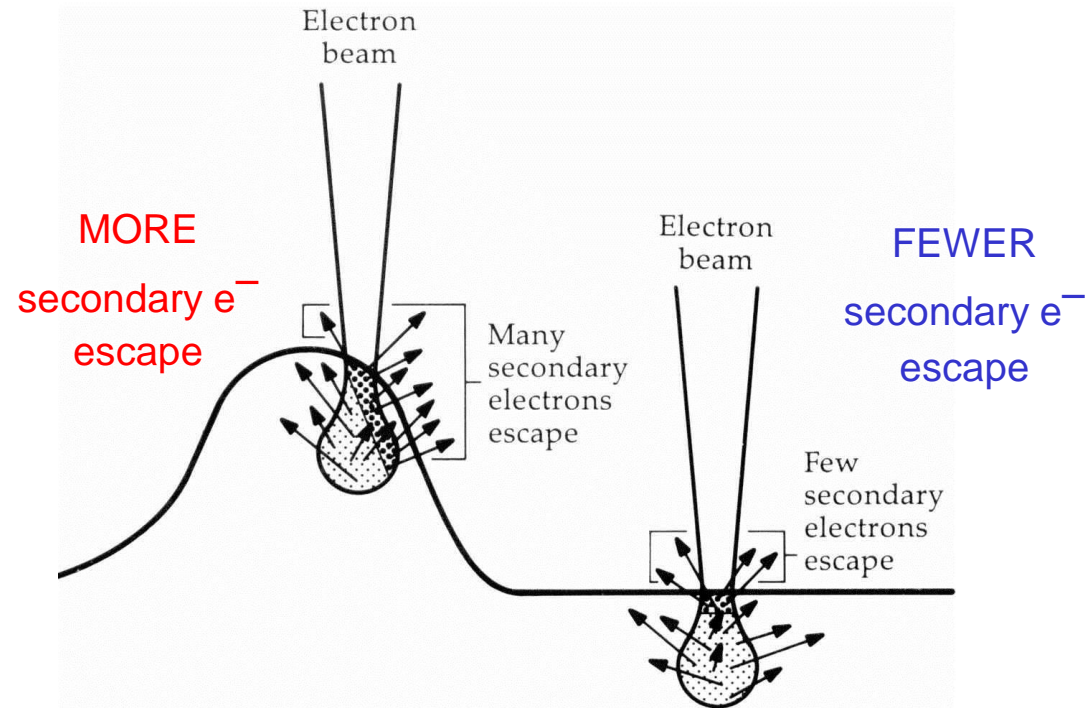
# Electron Microscopy



© JEOL, Ltd.

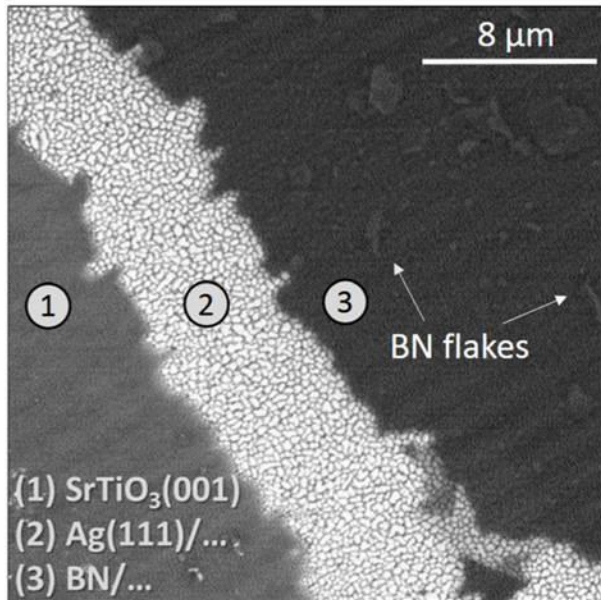
# Scanning Electron Microscopy (SEM)

- Secondary Electrons
  - Lost energy due to inelastic scattering
  - Sensitive to physical structure (topography)
- Backscattered Electrons
  - Elastically scattered
  - Sensitive to atomic number (elemental)
- X-ray Fluorescence
  - Collisionally excited states relax emitting X-rays

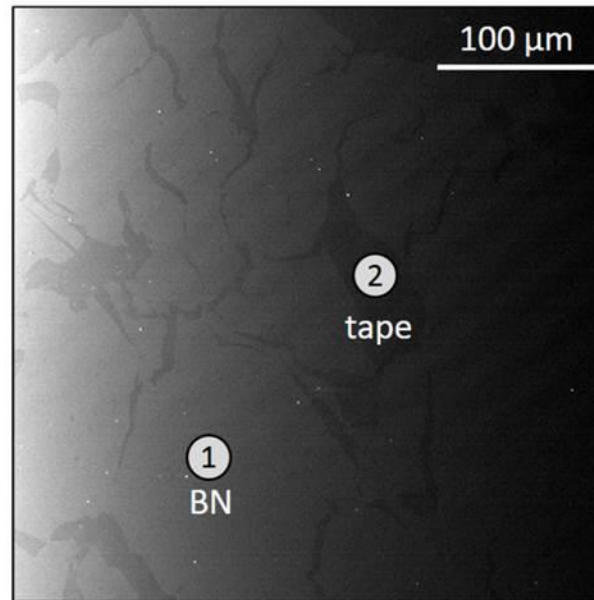


# SEM Boron Nitride

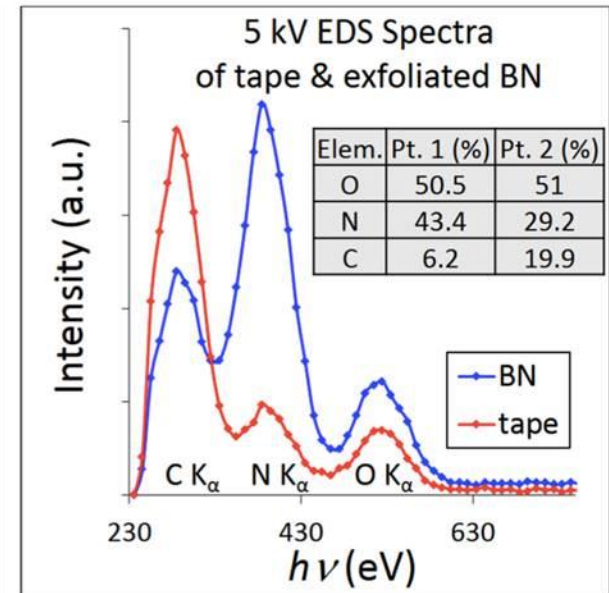
- Large BN flakes grown on thin film silver



(a) 5 kV image



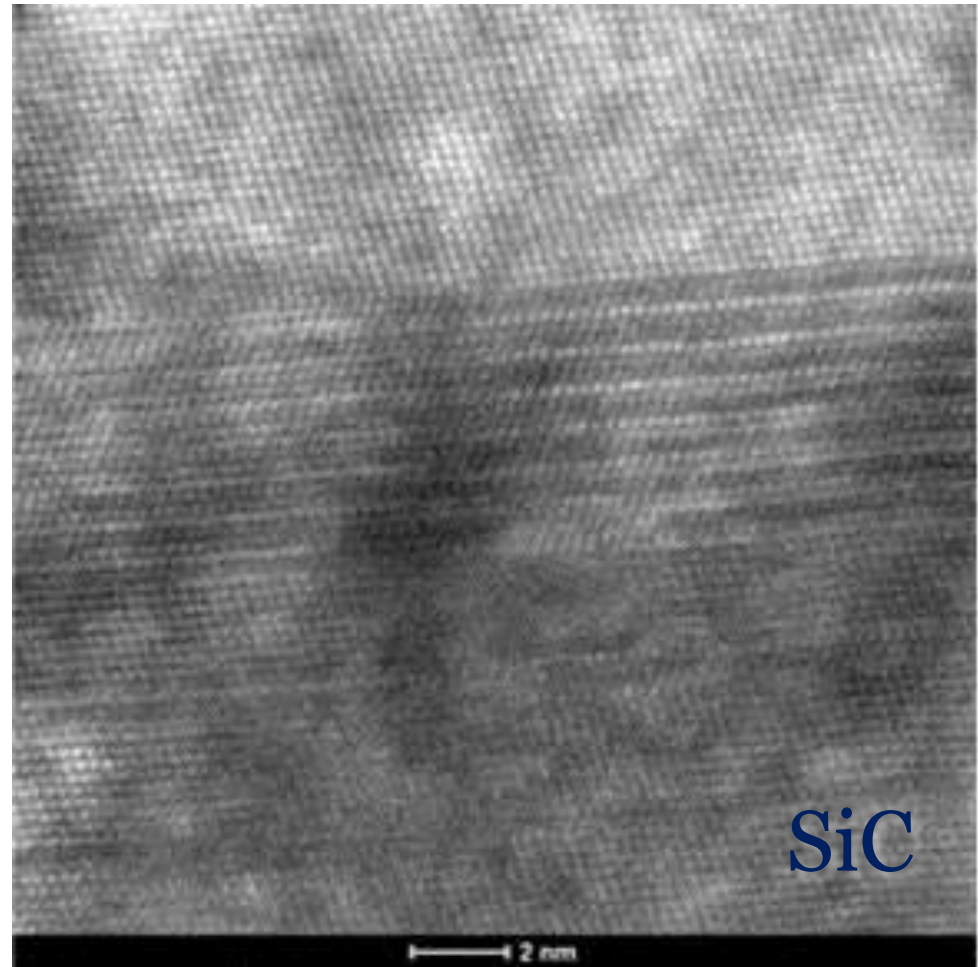
(b) 5 kV image



(c) 5 kV spectra

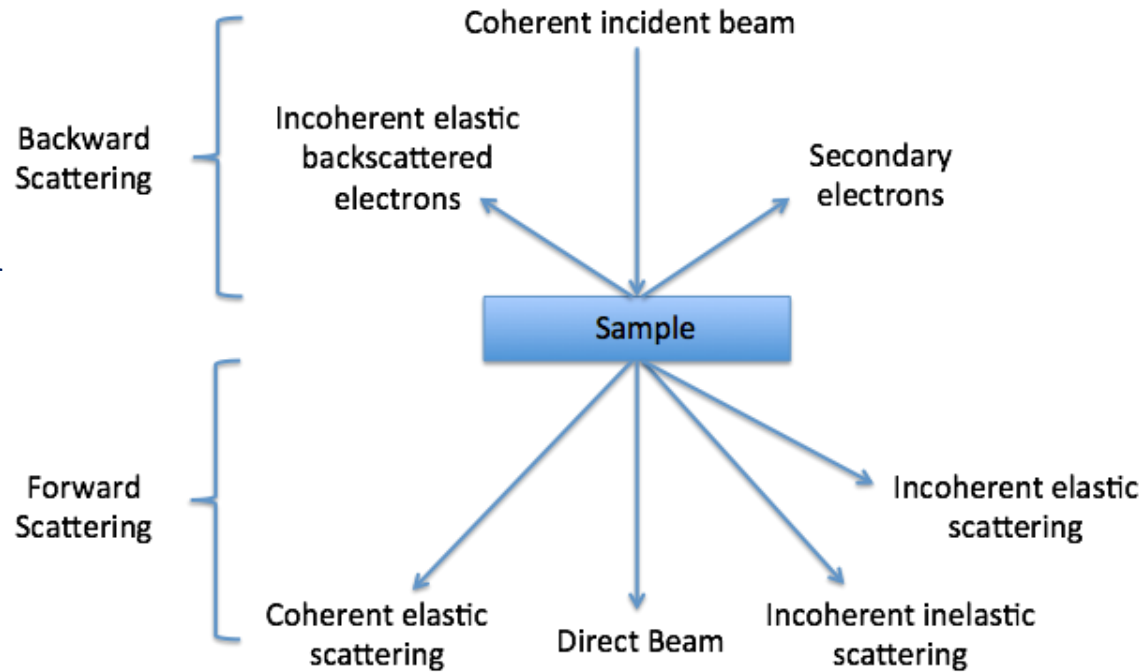
# Transmission Electron Microscopy (TEM)

- Topography
- Morphology
- Composition
- Crystallography



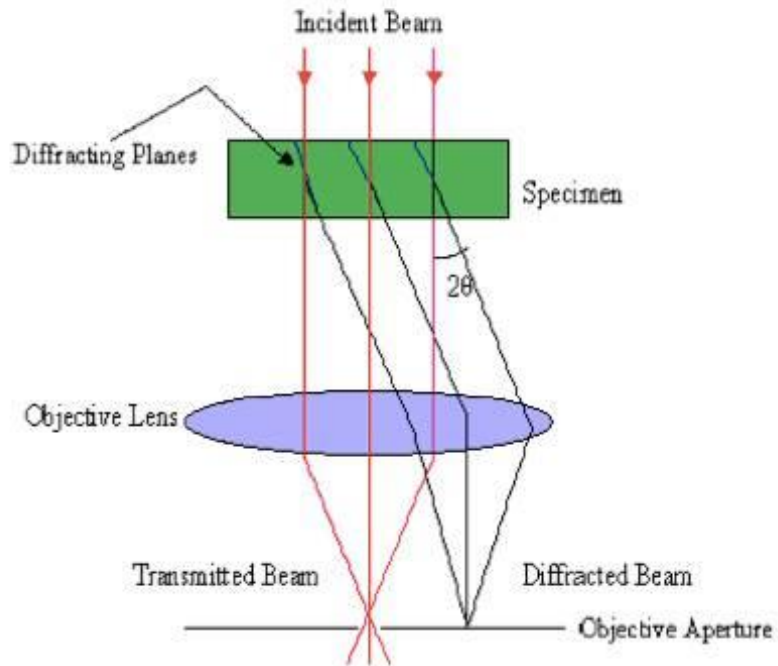
# TEM

- Thin (<100nm) specimen
- Electrons illuminate the specimen
- Objective lenses form an image or a diffraction pattern
- Electrons either pass through the sample unimpeded, or are scattered
- Image is viewed on a fluorescent screen

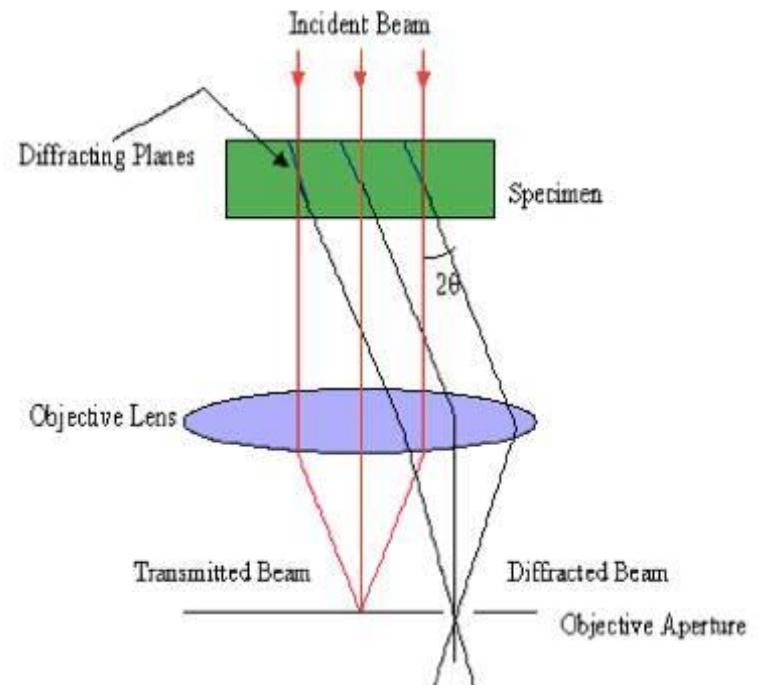




# TEM



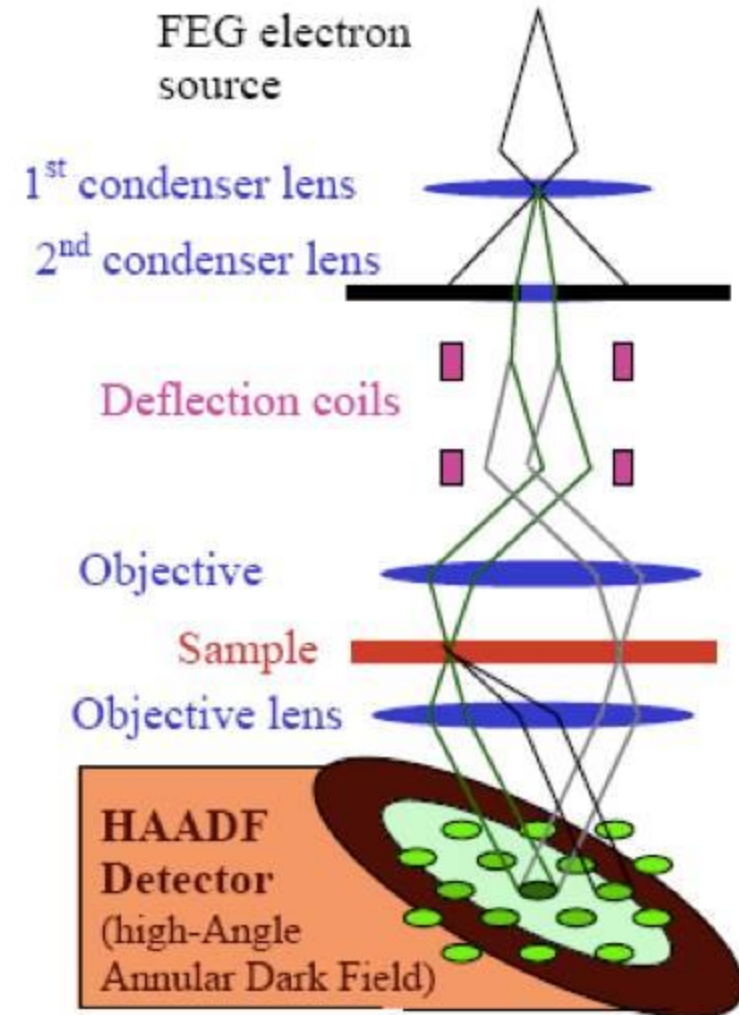
## Bright field



## Dark field

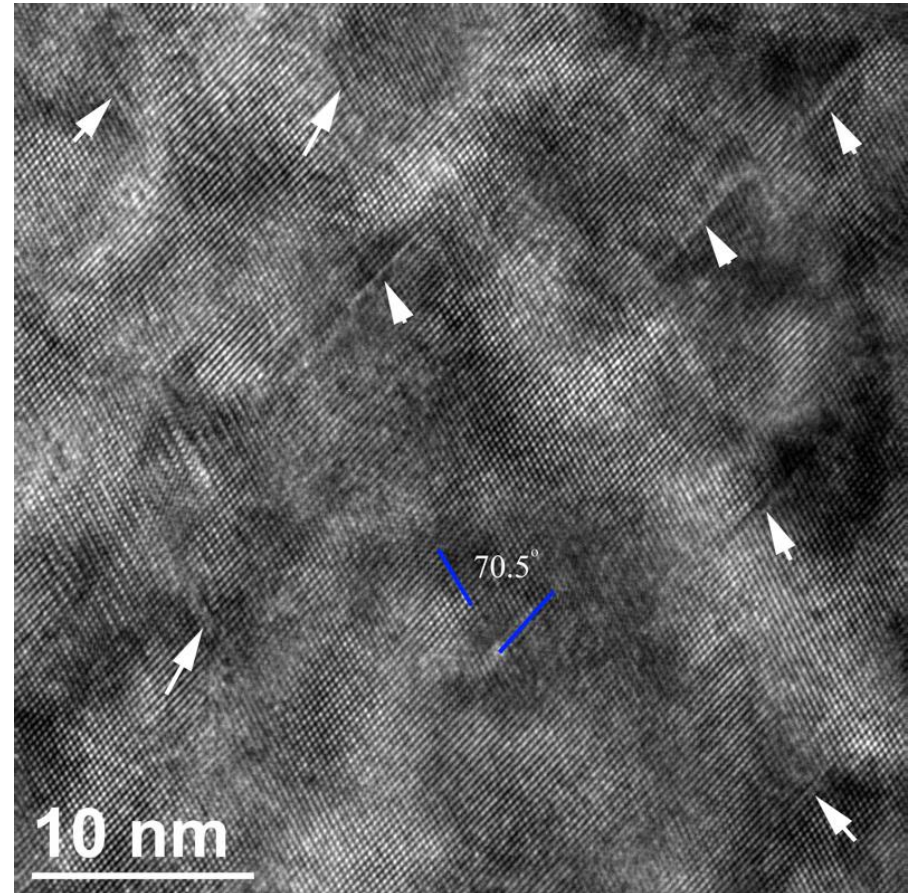
# High Angle, Annular Dark field (HAADF)

- High scattering angle
- Inelastic, incoherent electrons
- No diffraction



# TEM ZrC

- Dislocation loops
- TEM - proton irradiated 1.5 DPA @ 800 °C
- White arrows edge-on dislocation loops on (111) planes
- Angle between the edge-on dislocations on different (111) planes is  $\sim 70.5^\circ$

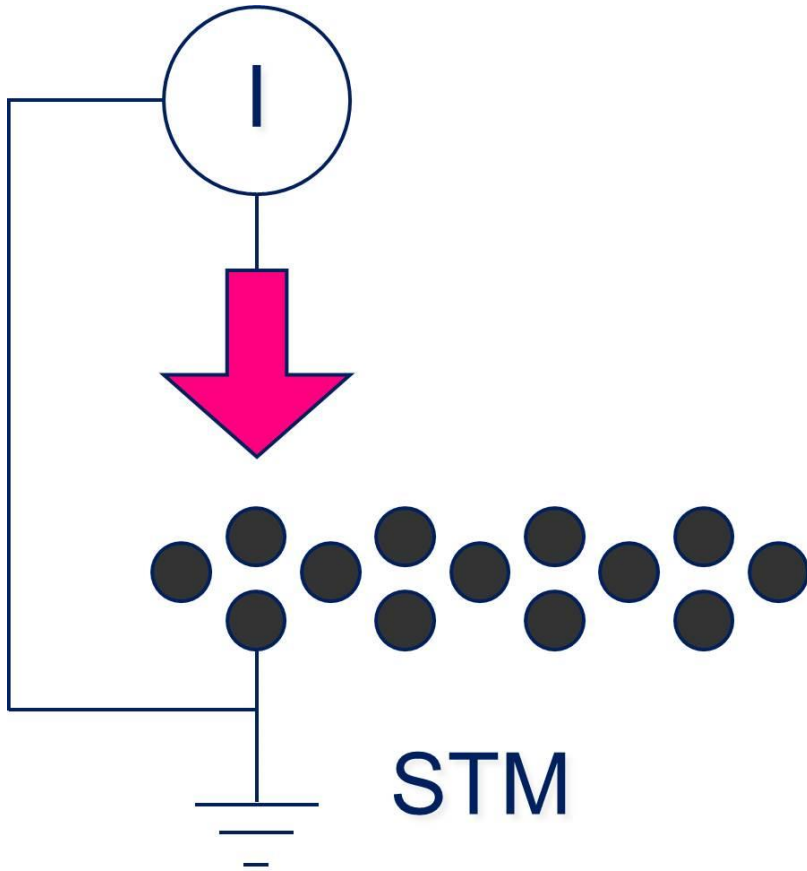


# Scanning Probe Microscopy

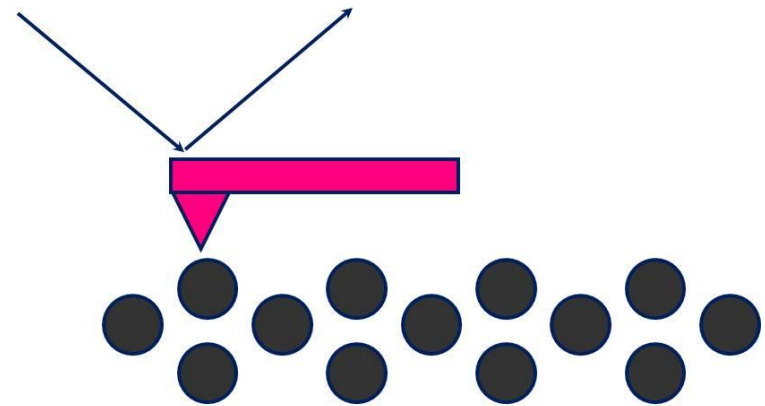
- Scanning Tunneling Microscopy
  - Tunneling current to tip
- Atomic Force Microscopy
  - Force on tip

# SPM

- Tip dragged across a sample

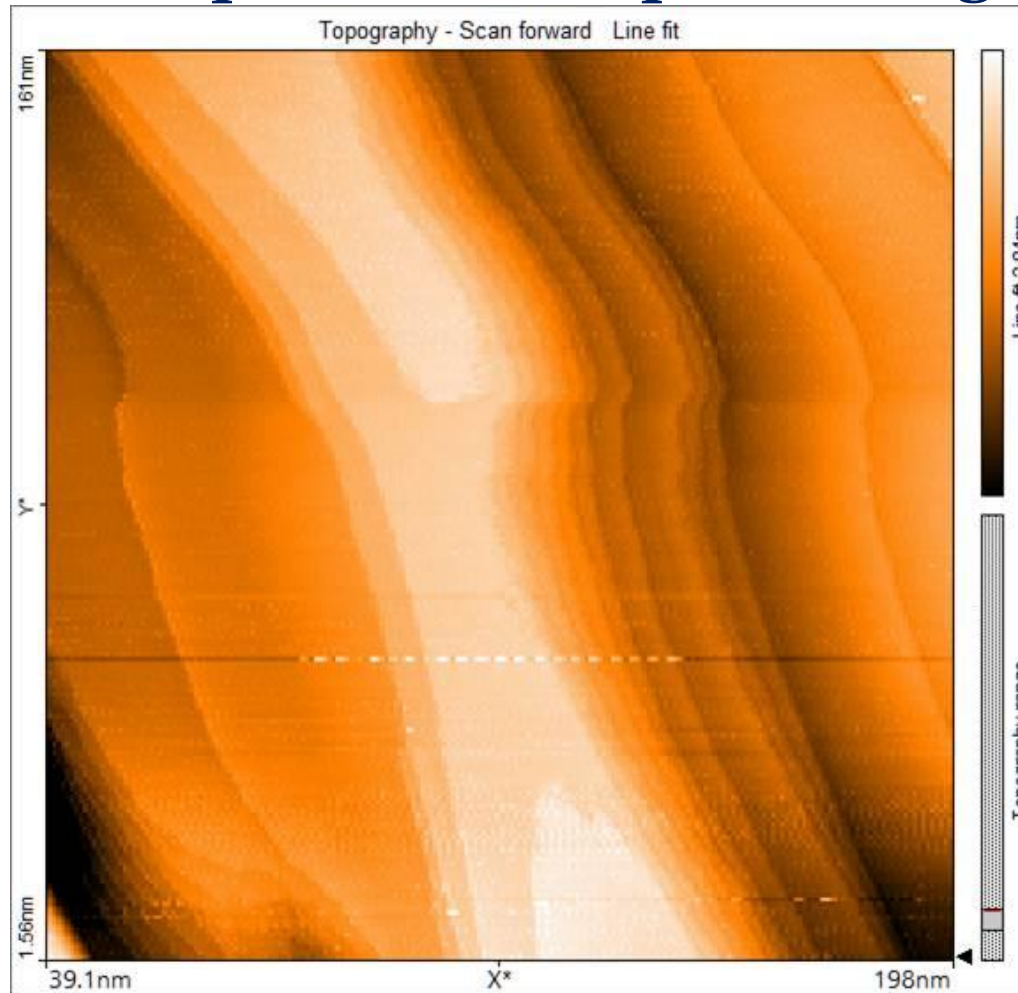


Photon



# Ag on Si(111)

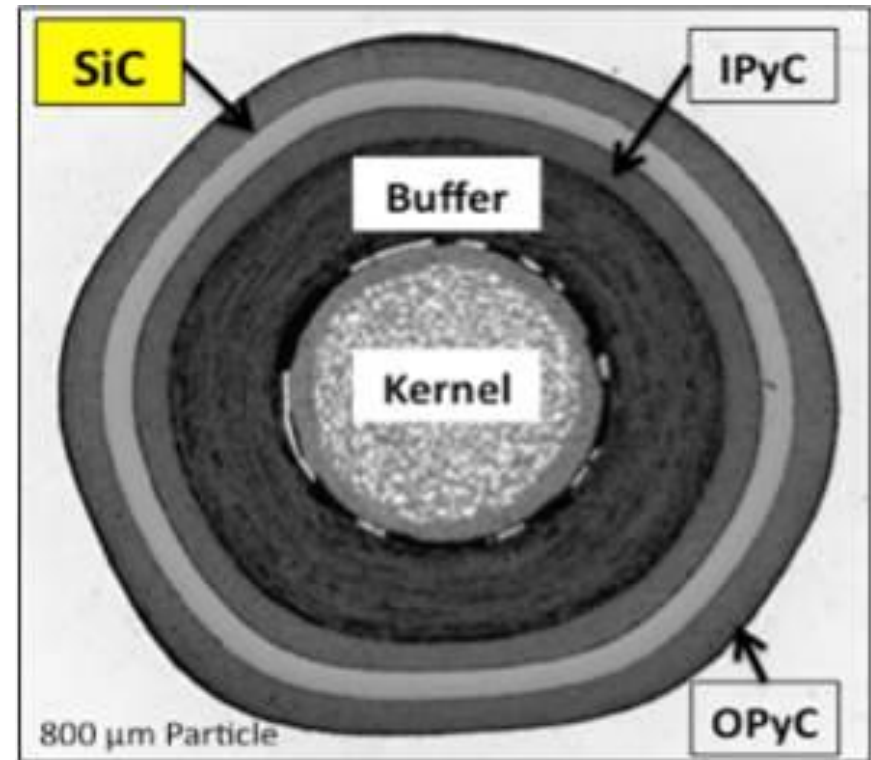
- Single atomic step islands of epitaxial Ag on Si(111)



Velazquez, et al., Applied Surface Science 360, 762 (2016)

# Putting Them All Together to Tell a Story

- Accident tolerant fuels
- High temperature reactors
  - TRISO Fuel



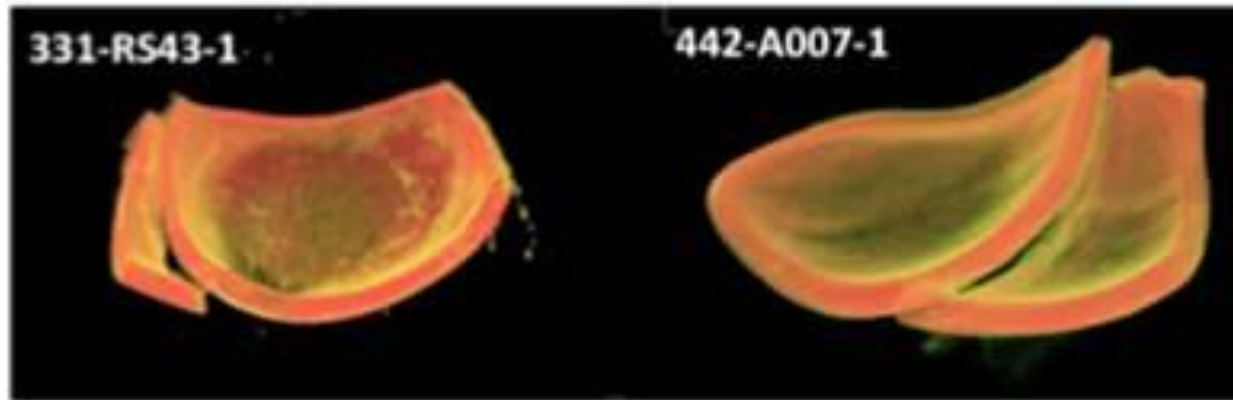
*As-fabricated TRISO fuel particle*

# Barrier Layer

- SiC is the barrier layer in TRISO fuels
- To get TRISO fuels qualified for use
  - Must understand mechanisms of damage



# Irradiated TRISO



X-ray Tomography of SiC fragments – C. Silva (ORNL)

- Irradiated in the Advanced Test Reactor
- After deconsolidation, select particles annealed to high temperatures for accident simulation
- Isolation of SiC shell through chemical processing

# Irradiated TRISO

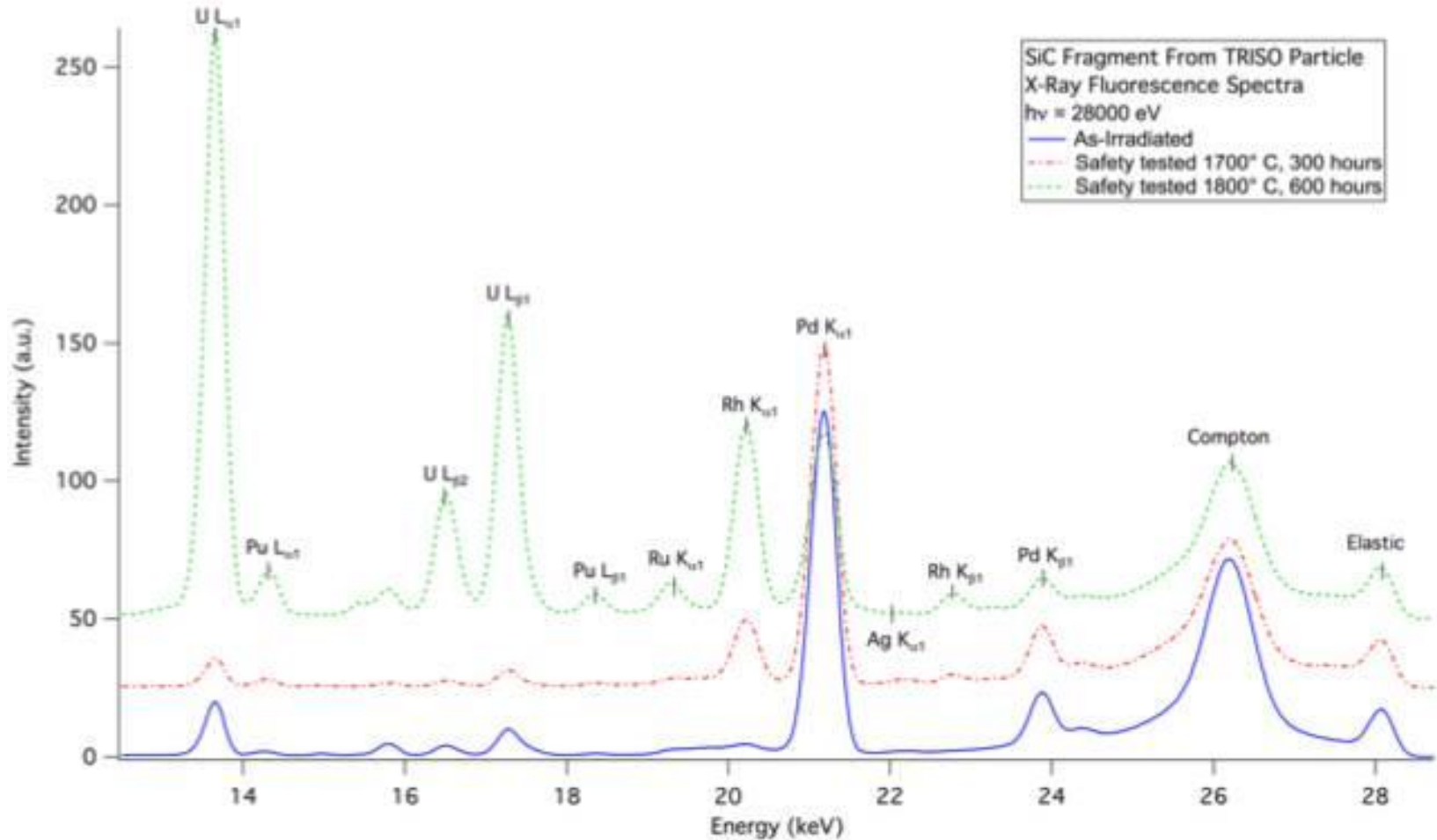
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<b>Parent Particle</b>	<b>Safety Testing</b>	<b>%FIMA</b>	<b>Ag Retention</b>
<b>611-AG59</b>	<b>None</b>	<b>15%</b>	<b>15%</b>
<b>331-RS43</b>	<b>300 hrs 1700 °C</b>	<b>19%</b>	<b>100% post-Test</b>
<b>422-A007</b>	<b>600 hrs 1800 °C</b>	<b>17%</b>	<b>Undetectable post-test</b>

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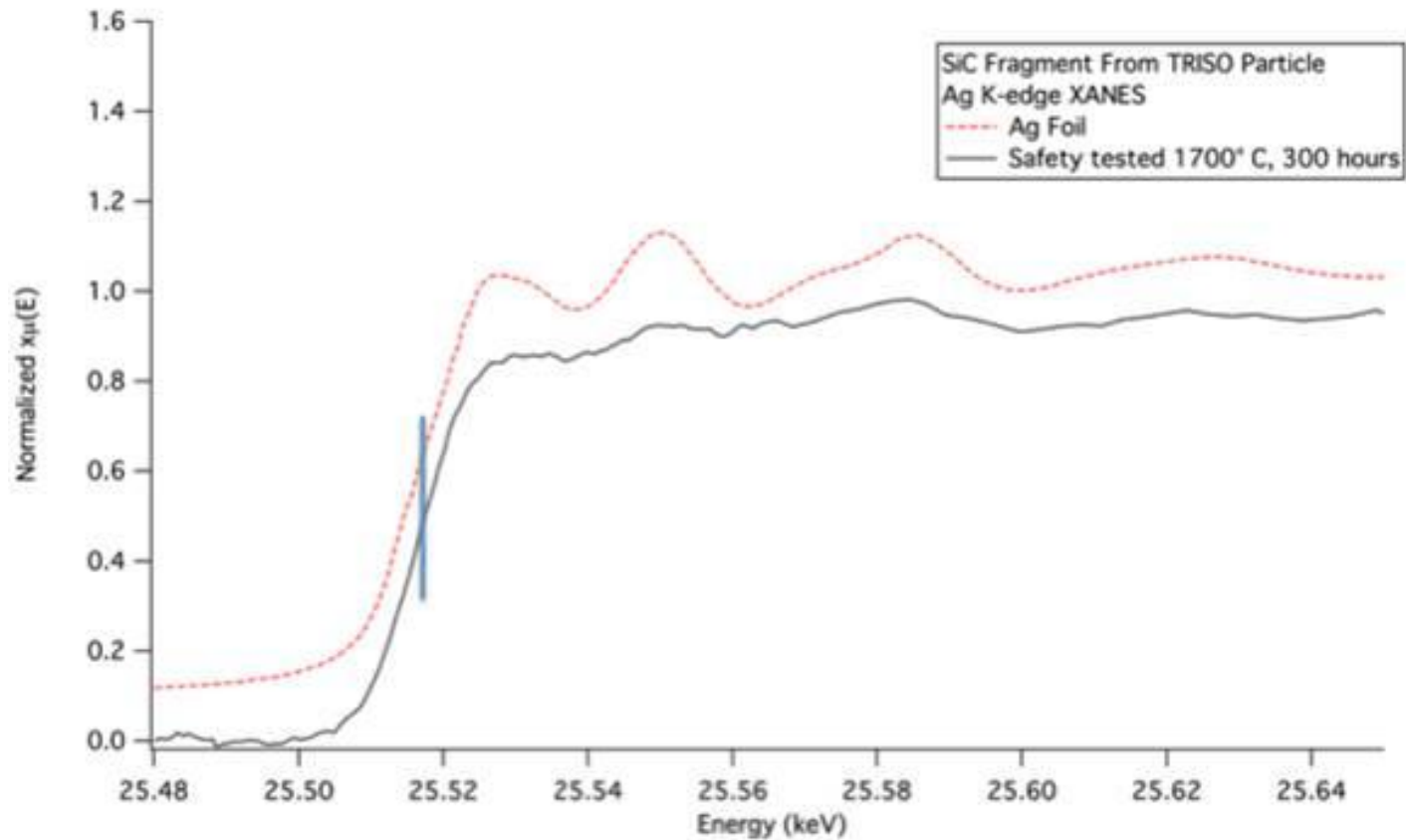
# Fluorescence

- Elemental analysis



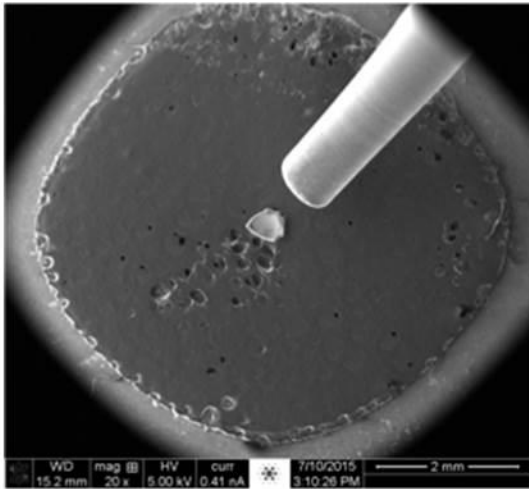
# XAS

- Ag is metallic
  - May be an alloy

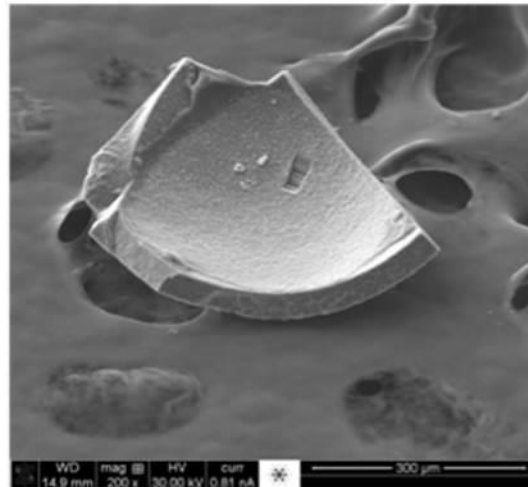


# Microscopy

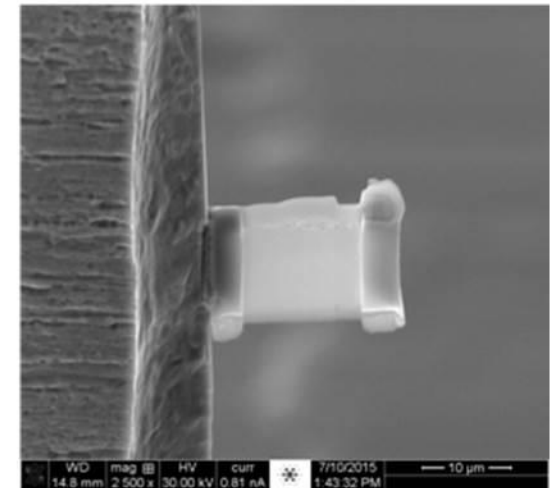
- Focused Ion Beam Milling



Particle is in the center of a 5mm carbon dot



Close-up of the particle curvature that must be worked with during sample preparation



Fully thinned specimen prepared for TEM analysis

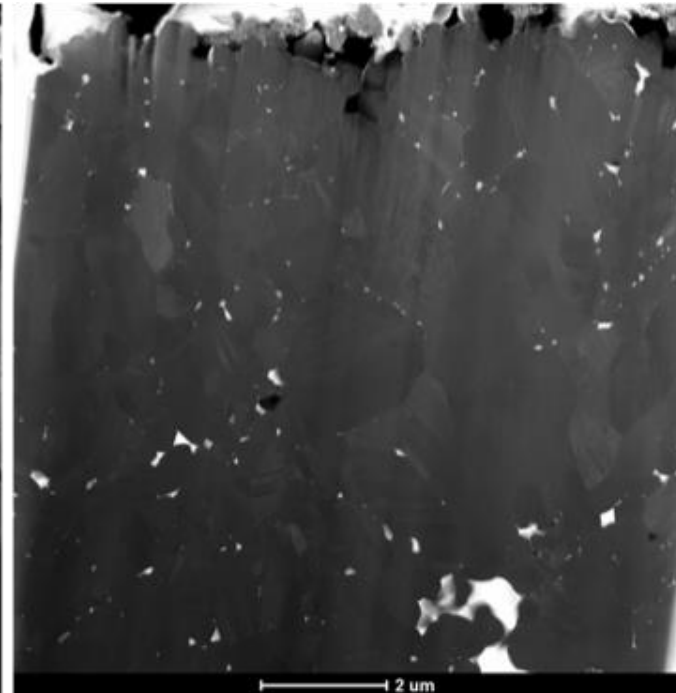
# As-Irradiated Sample: IPyC/SiC Interface

- High angle, annular dark field shows high Z

## Bright Field

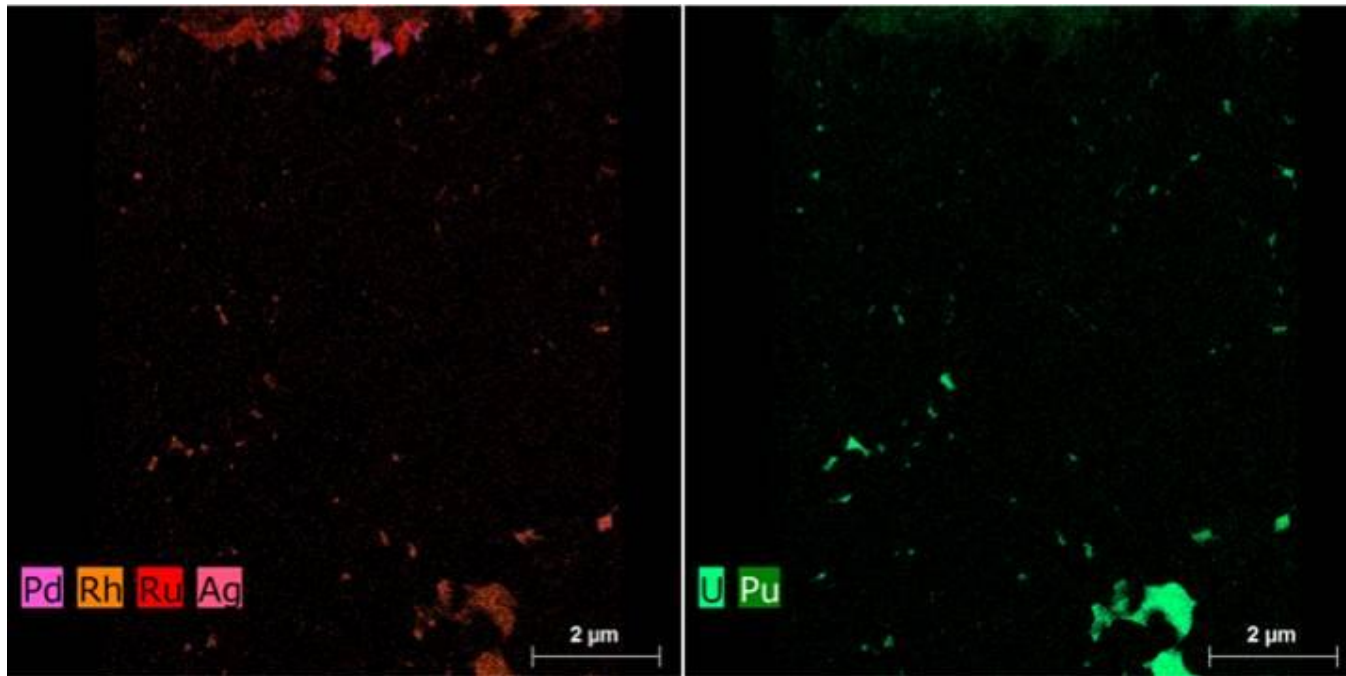


## HAADF



# As-Irradiated Sample: IPyC/SiC Interface

- Lot of overlap among fission products
- Some co-location with actinides



Fission

Actinides

# What Did We Learn By Combining Techniques?

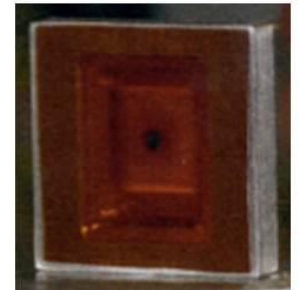
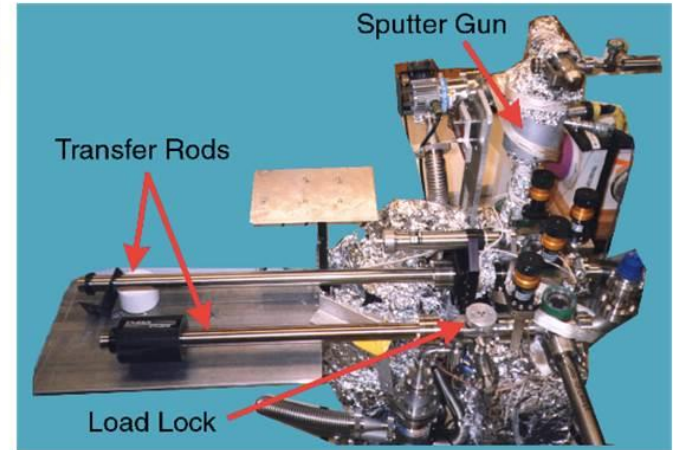
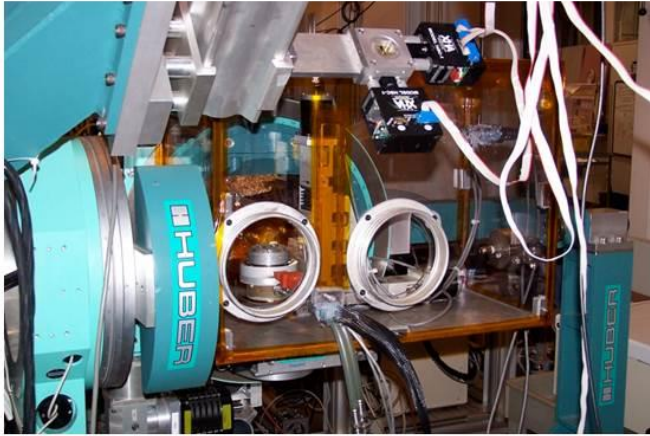
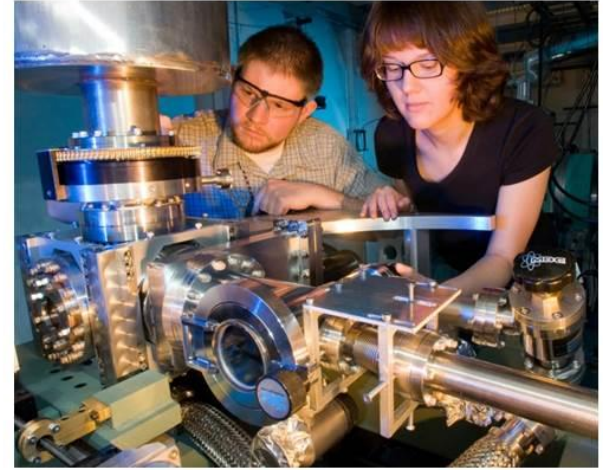
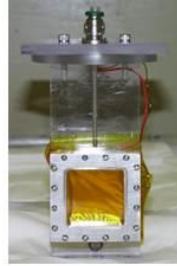
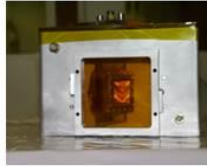
- Fission products and actinides found along grain boundaries and triple points in the SiC.
- Fission products (specifically Pd) found all the way to the OPyC interface
- U has been found in large quantities as deep as 13 $\mu$ m from the IPyC interface, and in trace quantities near the OPyC interface
- Ag is metallic



# Safety Concerns

- Working with radioactive materials does require extra precaution
- Minimize worker exposure
  - Use long tools
  - Redesign holders to minimize handling time

# Safety



Terry, unpublished

# Nuclear Science User Facility

- Almost all of these techniques are available to all through the Nuclear Science User Facility
  - Write a proposal
  - If awarded, assigned Subject Matter Expert to work with you
  - Great way to get access to these techniques



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- Seibert, et al., in preparation
- C. Silva, ORNL

# Students and Postdocs

- Dan Olive
- Hasitha Ganegoda
- Daniel Velazquez
- Joshua Wright
- Hamdi Man
- Rachel Seibert
- ZhengRong Lee

Questions???

# Upcoming Webinars

- Nuclear Materials Analysis — Chemical Methods
- Nuclear Materials Analysis — Non-Destructive Analysis
- Nuclear Materials Analysis — Mass Spectroscopy

NAMP website: [www.wipp.energy.gov/namp](http://www.wipp.energy.gov/namp)